

A. Dunand<sup>1</sup>, M. Ialovega<sup>1,2</sup>, E.A. Hodille, C. Martin<sup>1</sup>, C. Pardanaud<sup>1</sup>, M. Minissale<sup>1</sup>, E. Bernard<sup>2</sup>, T. Angot<sup>1</sup>, C Grisolia<sup>2</sup> and R Bisson<sup>1</sup>

<sup>1</sup>*Aix-Marseille Univ, CNRS, PIIM, UMR 7345, Marseille, France*

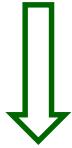
<sup>2</sup> *CEA, IRFM, F-13108, Saint Paul-lez-Durance, France*



*PIIM laboratory  
Aix-Marseille University - CNRS  
Marseille, France*

# The effect of oxygen in the bulk of tungsten on deuterium retention: a fundamental approach

1. D implantation in W samples (IB)



2a. D retention (TPD, NRA coll. JSI)

2b. W characterization (FIB-SEM, AES, LEED, XPS)

experiments

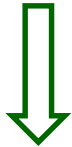
UHV:  $10^{-10}$  mbar

ion gun:  $10^{16}$ - $10^{18}$  ion.m<sup>-2</sup>.s<sup>-1</sup>

TPD: 1 – 10 K.s<sup>-1</sup>

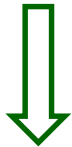
# The effect of oxygen in the bulk of tungsten on deuterium retention: a fundamental approach

1. D implantation in W samples (IB)

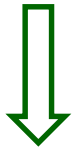


2a. D retention (TPD, NRA coll. JSI)

2b. W characterization (FIB-SEM, AES, LEED, XPS)



3. Macroscopic Rate Equations model (MRE)  
(predict  $^3\text{H}$  retention in fusion reactor)  
initialized by Density Functional Theory (DFT)



4. Understand fundamental  
D-W interaction

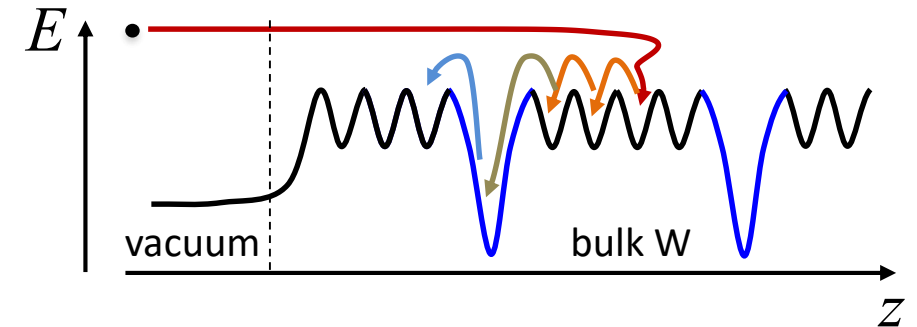
experiments

modeling

UHV:  $10^{-10}$  mbar

ion gun:  $10^{16}$ - $10^{18}$  ion.m $^{-2}$ .s $^{-1}$

TPD: 1 – 10 K.s $^{-1}$



ion implantation

bulk diffusion (Fick)

$$\frac{\partial c_m}{\partial t} = \varphi \cdot (1 - r) \cdot f(z) - v_{\text{diff}} \cdot \frac{\partial^2 c_m}{\partial z^2} - \frac{\partial c_t}{\partial t} \quad (1)$$

$$\frac{\partial c_t}{\partial t} = v_{\text{trap}} \cdot (c_m/n_m) \cdot (n_t - c_t) - v_{\text{detrap}} \cdot c_t \quad (2)$$

trapping

and detrapping at a defect

$$v_{\text{process}} = v_0 \times e^{\left(-\frac{E_a}{k_b T}\right)}$$

# Deuterium retention in tungsten in polycrystalline W experimental dataset to guide a DFT-MRE model

1. D implantation (IB)



2a. D retention (TPD, NRA)

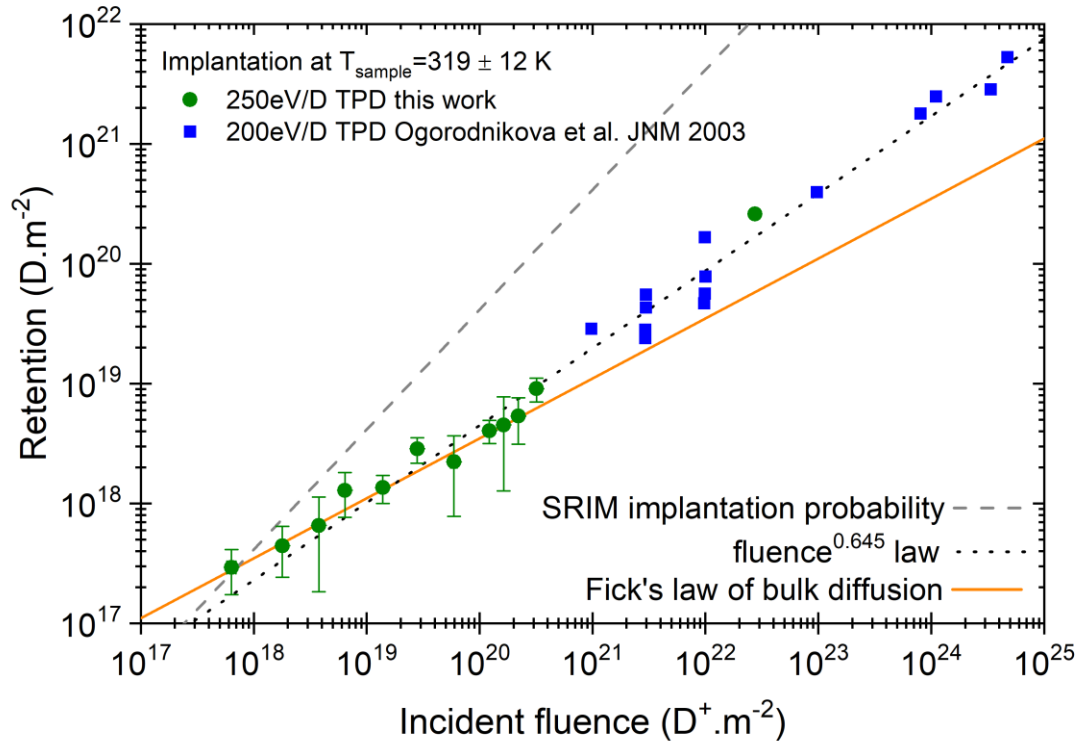
2b. W characterization  
(FIB-SEM, AES, XPS)



3. DFT-MRE modeling



4. D – W interaction



➤ D retention in polycrystalline W does not follow Fick's law of bulk diffusion

➤ Higher D retention implies defect(s) trapping (bulk and/or near-surface)

Bisson *et al.*, J. Nucl. Mater. **467** (2015) 432

# Deuterium retention in tungsten in polycrystalline W experimental dataset reproduced by a DFT-MRE model

1. D implantation (IB)



2a. D retention (TPD, NRA)

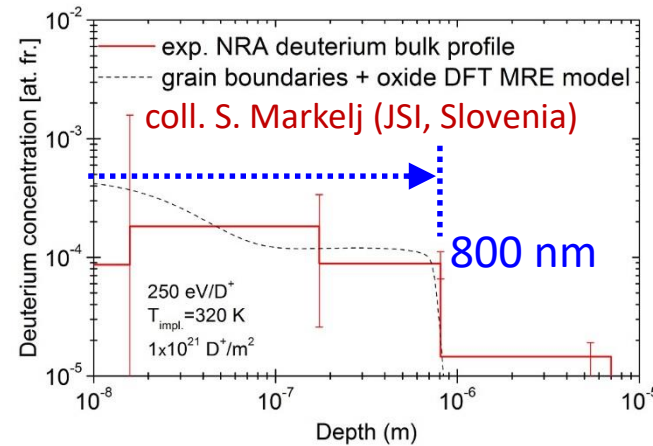
2b. W characterization  
 (FIB-SEM, AES, XPS)



3. DFT-MRE modeling  
 (oxide + GB)



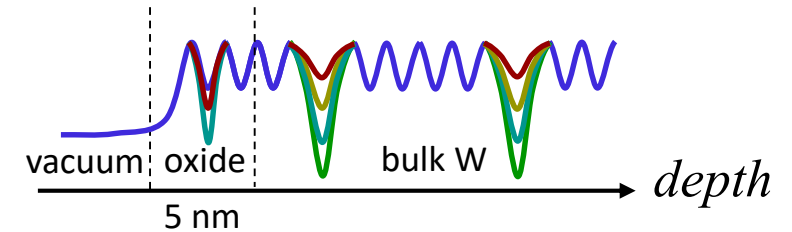
4. D – W interaction



NRA deuterium depth profile in poly-W

→ near surface trapping + bulk trapping

« native oxide » (AES) + grain boundaries (FIB-SEM)



■ DFT detrapping energies → drive isothermal desorption

- Oxygen – W vacancy cluster (native oxide)

Kong *et al.* J. Nucl. Mater. 433 (2013) 357

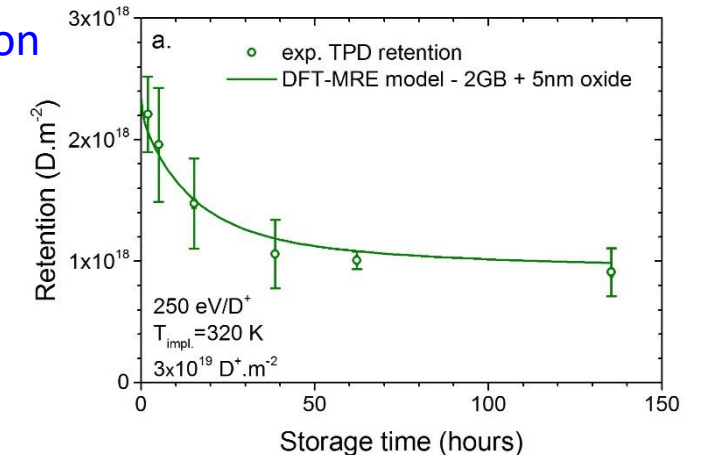
- Grain boundaries (GB)

Xiao *et al.* J. Nucl. Mater. 430 (2012) 132

■ 3 free parameters → quantitative agreement

2 densities: native oxide + GB

native oxide thickness (guided by AR-XPS)



Hodille *et al.*, Nuclear Fusion **57** (2017) 076019

# Deuterium retention in tungsten in polycrystalline W experimental dataset reproduced by a DFT-MRE model

1. D implantation (IB)



2a. D retention (TPD, NRA)

2b. W characterization

(FIB-SEM, AES, XPS)

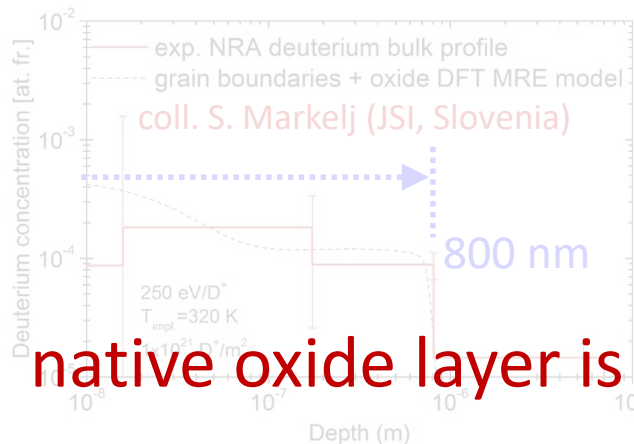


3. DFT-MRE modeling

(oxide + GB)



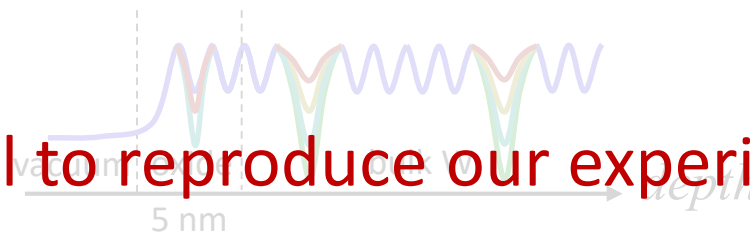
4. D – W interaction



NRA deuterium depth profile in W

→ near surface trapping + bulk trapping

« native oxide » (AES) + grain boundaries (FIB-SEM)



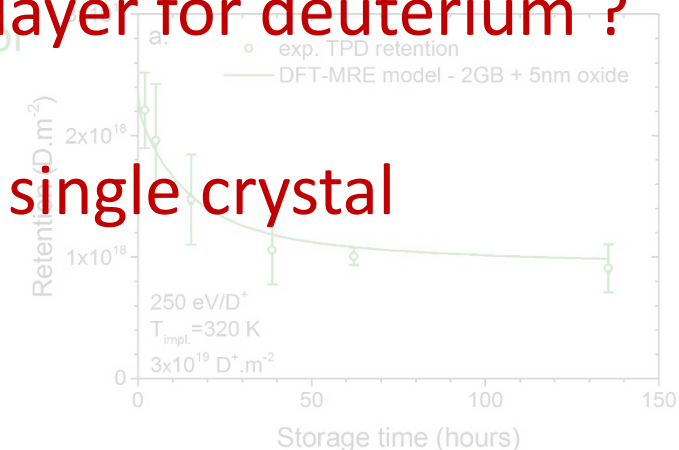
• native oxide layer is essential to reproduce our experiments

• is W native oxide really a trapping layer for deuterium?

- DFT detrapping energies → drive both the desorption and the adsorption
  - Oxygen – W vacancy cluster (native oxide)  
 Kong *et al.* J. Nucl. Mater. 433 (2013) 357
  - Oxygen – W dislocation core  
 Xiao *et al.* J. Nucl. Mater. 430 (2012) 132

➤ looking for a direct evidence on W single crystal

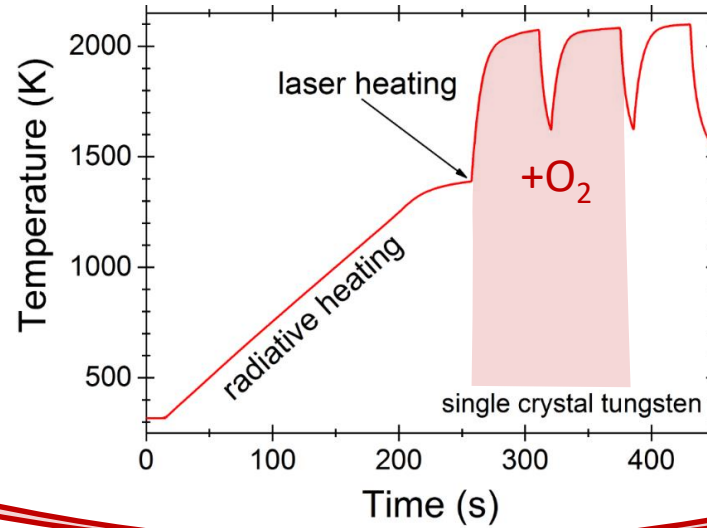
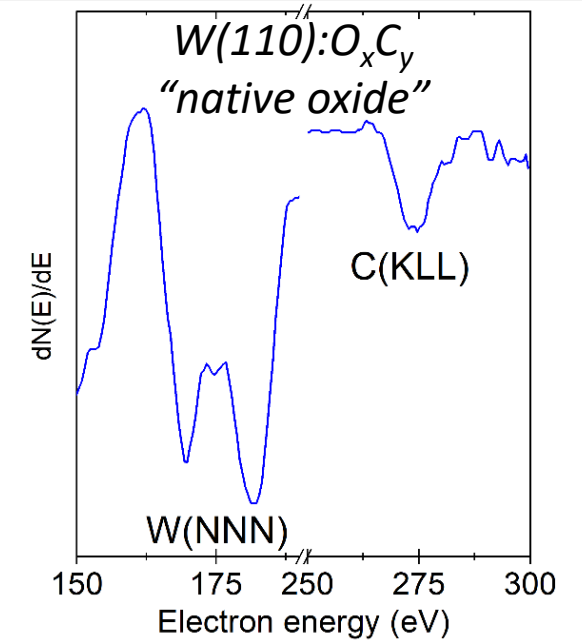
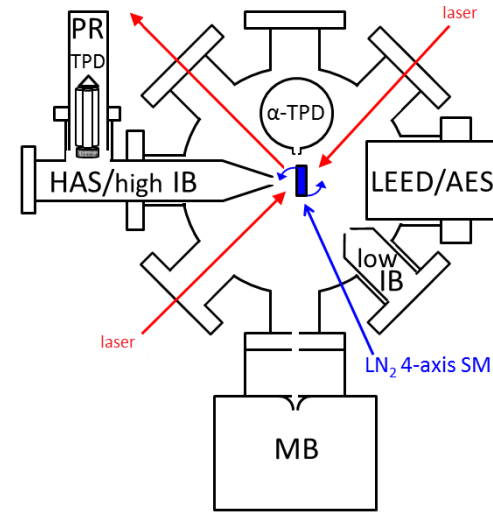
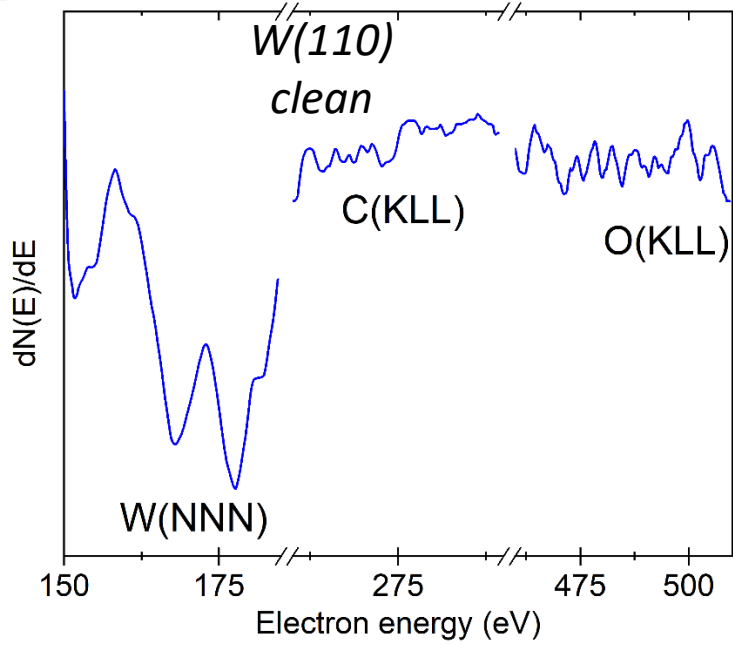
- 3 free parameters → quantitative agreement
  - 2 densities: native oxide + GB
  - native oxide thickness (guided by AR-XPS)



Hodille *et al.*, Nuclear Fusion 57 (2017) 076019



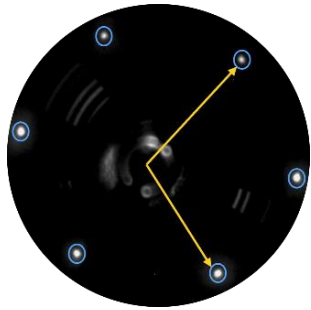
# Deuterium retention in single crystal tungsten W(110) – is the native oxide a trapping layer ?



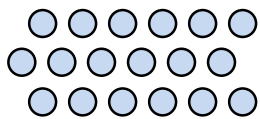
# Deuterium retention in single crystal tungsten

## W(110) – is the native oxide a trapping layer ?

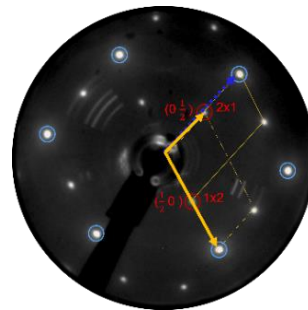
W(110)  
clean



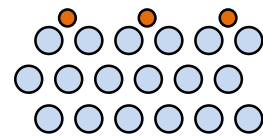
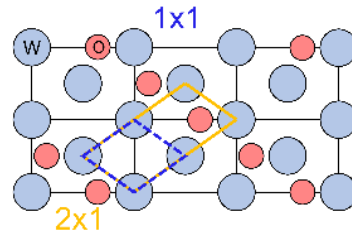
- LEED: 1x1
- structure of clean W(110)
- AES: only W



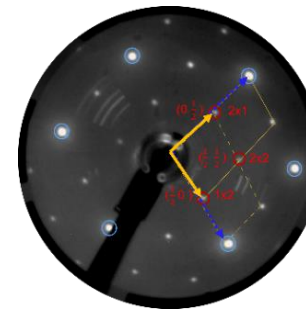
W(110):O<sub>0.50ML</sub>  
(2x1)



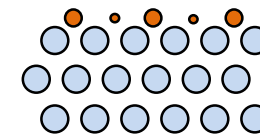
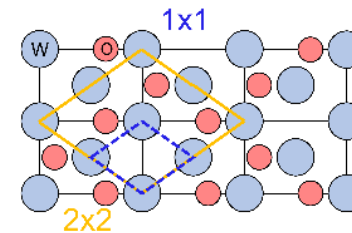
- LEED: 2x1
- ~0.50 ML



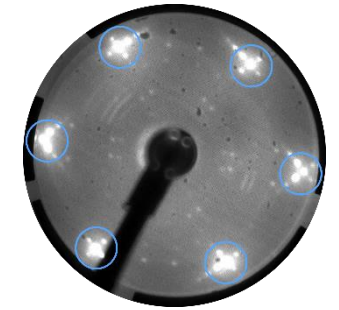
W(110):O<sub>0.75ML</sub>  
(2x2)



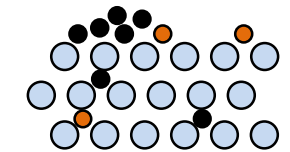
- LEED: 2x2
- ~0.75 ML



W(110):O<sub>x</sub>C<sub>y</sub>  
"native oxide"



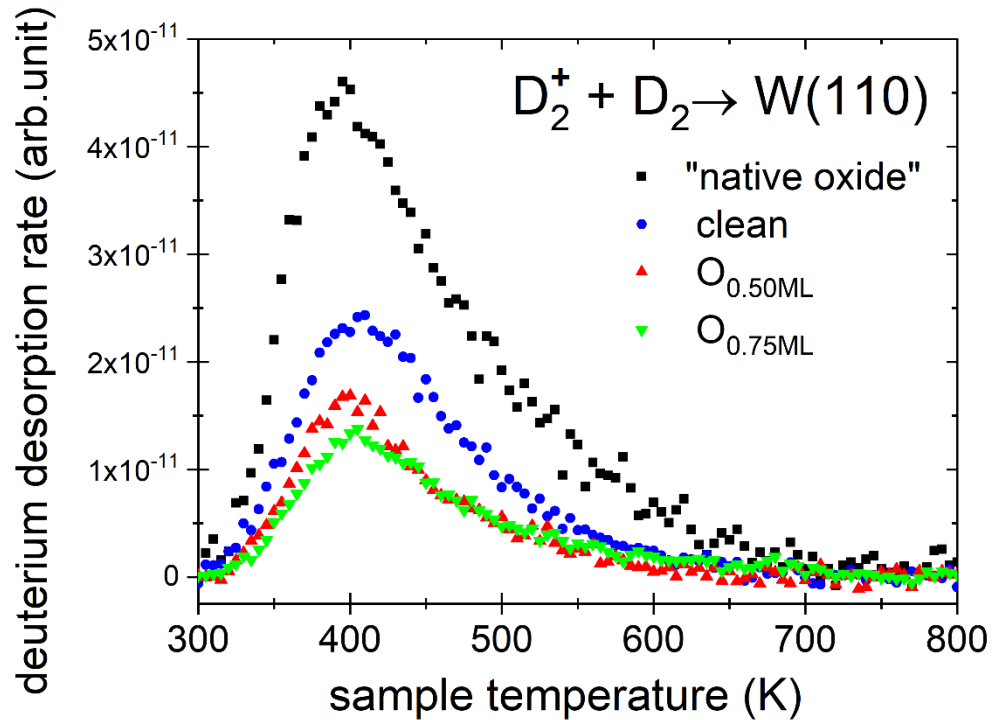
- LEED: crystalline structures + amorphous background
- AES: presence of C and O in the "native oxide"





# Deuterium retention in single crystal tungsten

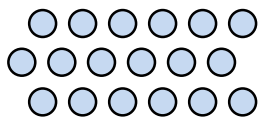
## W(110) – is the native oxide a trapping layer ?



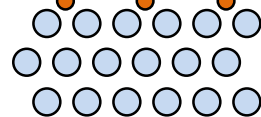
- ✓ removing the “native oxide” reduces D retention
- ! adding a sub-monolayer “oxide” reduces D retention !?!
- ✓ here, we have both  $D_2^+$  implantation and residual  $D_2$
- D retention can originate from both bulk & surface

Dunand *et al.*, Nuclear Fusion **65** (2022) 054002

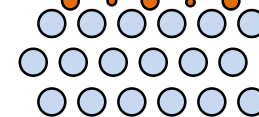
clean



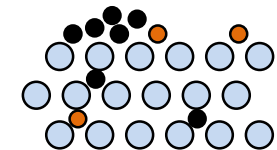
$O_{0.50ML}$



$O_{0.75ML}$

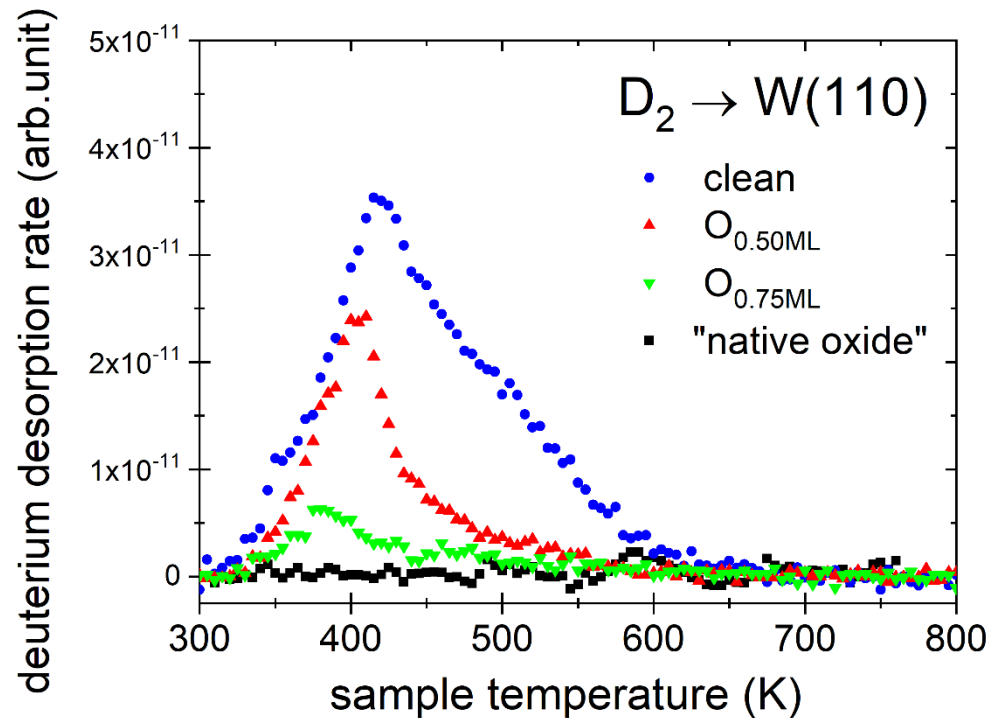


“native oxide”



# Deuterium retention in single crystal tungsten

## W(110) – is the native oxide a trapping layer ?



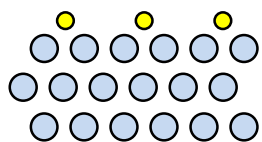
- ✓ the "native oxide" forbids D retention on the W surface
- ✓ the clean W surface and sub-monolayers of O exhibit D surface retention

consistent with Whitten & Gomer Surf. Sci. **409** (1998) 16

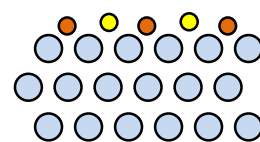
- subtract this D adsorption from TPD of  $D_2^+$  implantation to estimate roughly the bulk retention significance

Dunand *et al.*, Nuclear Fusion **65** (2022) 054002

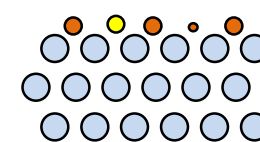
clean



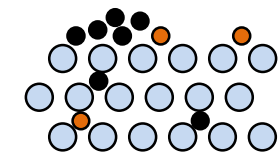
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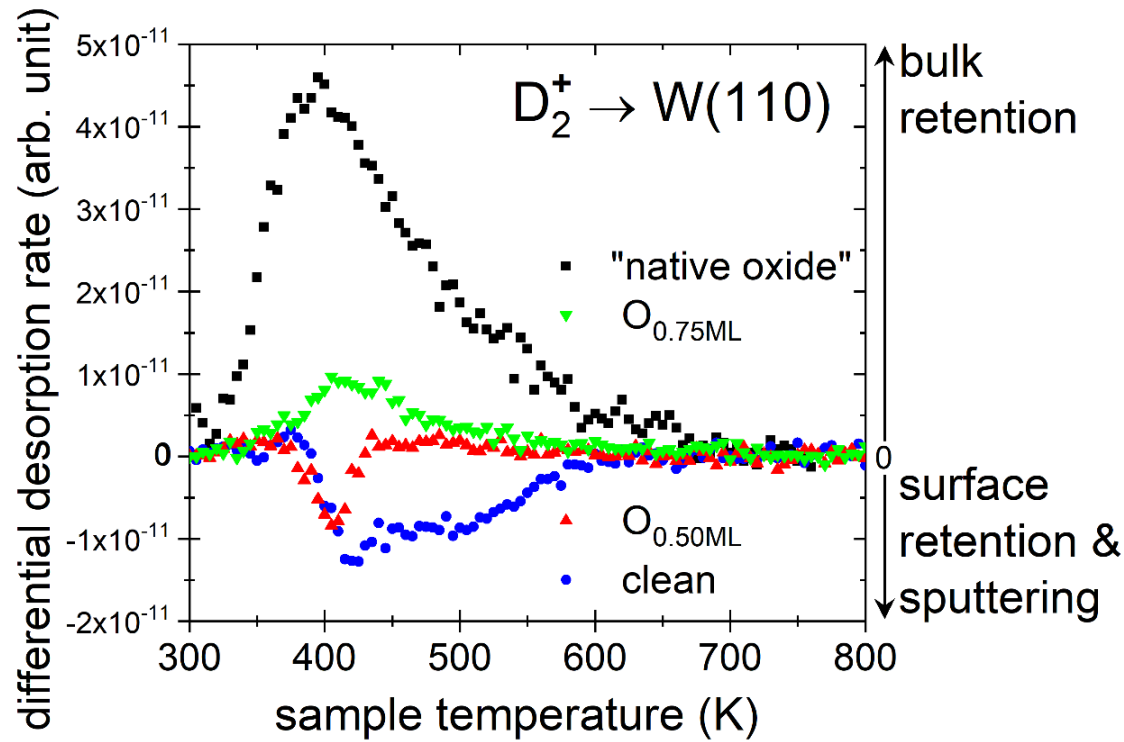


"native oxide"



# Deuterium retention in single crystal tungsten

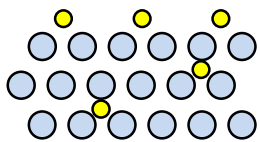
## W(110) – is the native oxide a trapping layer ?



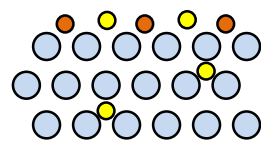
- ✓ for  $O \leq 0.50$  ML: D surface retention is significant and D sputtering plays a role
- ✓ for  $O \geq 0.75$  ML bulk trapping is preponderant
- Native oxide a “bulk (near-surface)” trapping layer... but it contains carbon impurities
- ➔ grow pure thick oxide to probe the effect of oxygen only

Dunand *et al.*, Nuclear Fusion **65** (2022) 054002

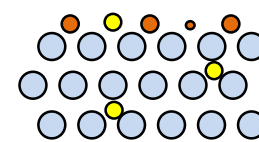
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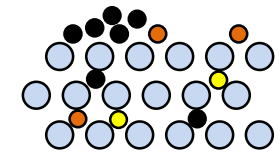
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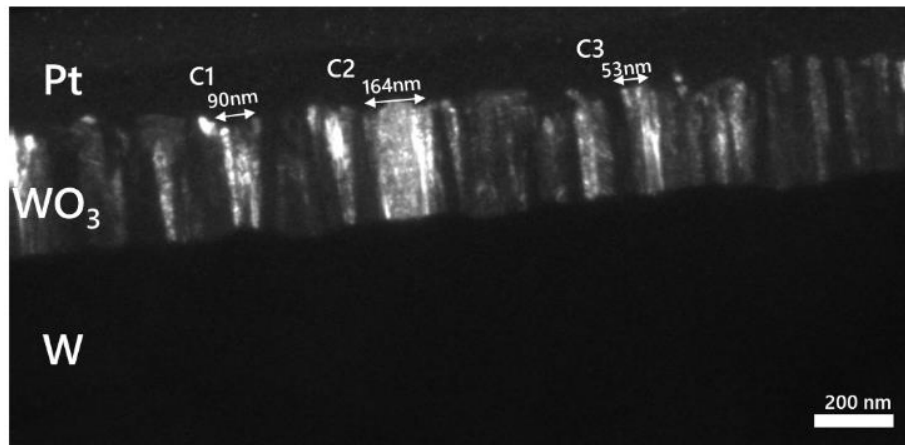
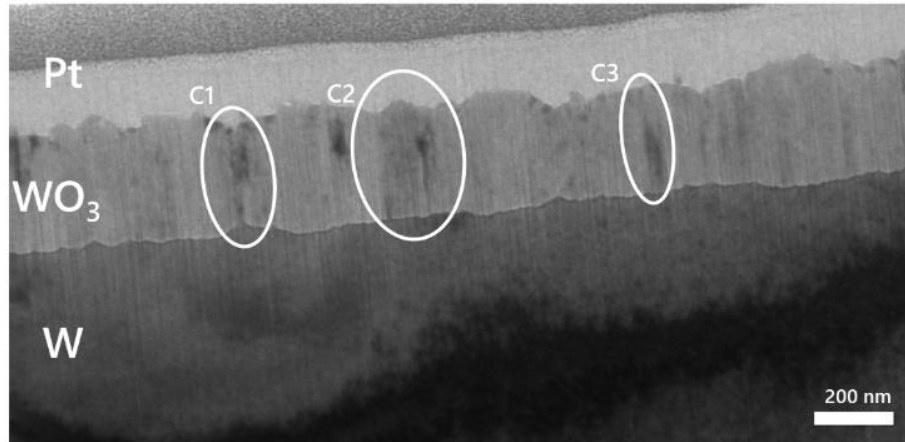


“native oxide”



# Deuterium retention in bulk polycrystalline tungsten oxide

$WO_3$  – is pure bulk oxide also a trapping layer ?



TEM

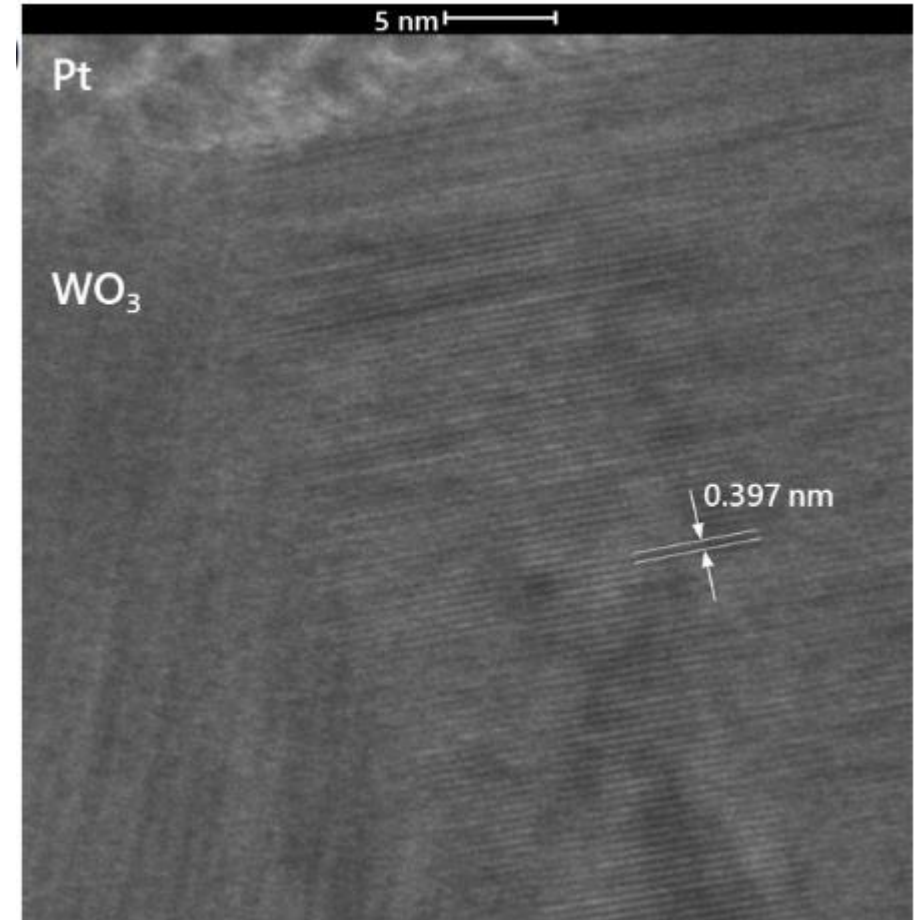
top/bottom: bright/dark field

200 nm thick WO<sub>3</sub>  
with columnar  
structure

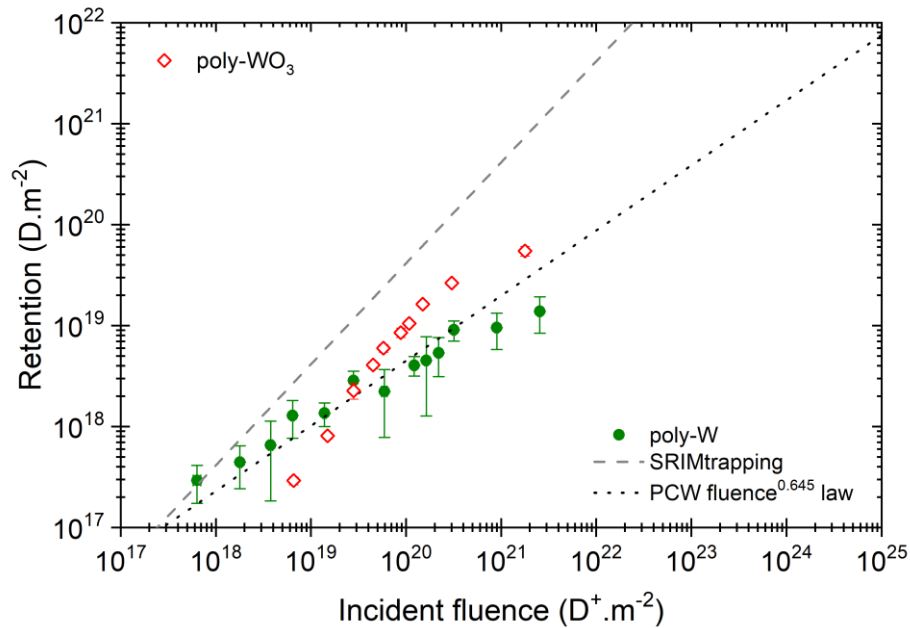
SRIM D<sub>2</sub><sup>+</sup>  
implantation range  
of about 20 nm

Repeated D<sub>2</sub><sup>+</sup>  
implantation/TPD  
on a single sample  
possible since this  
WO<sub>3</sub> is thermally  
stable up to ~800 K

lalovega *et al.*, in preparation



HRTEM



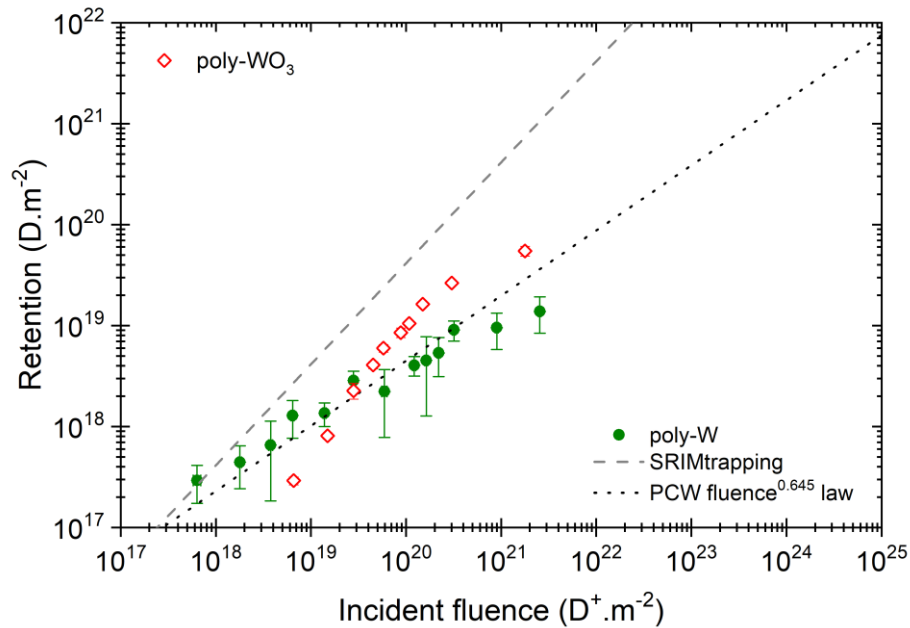
200 nm thick  
WO<sub>3</sub> stable up  
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- Low fluence = lower D retention vs poly-W
- High fluence = higher D retention vs poly-W

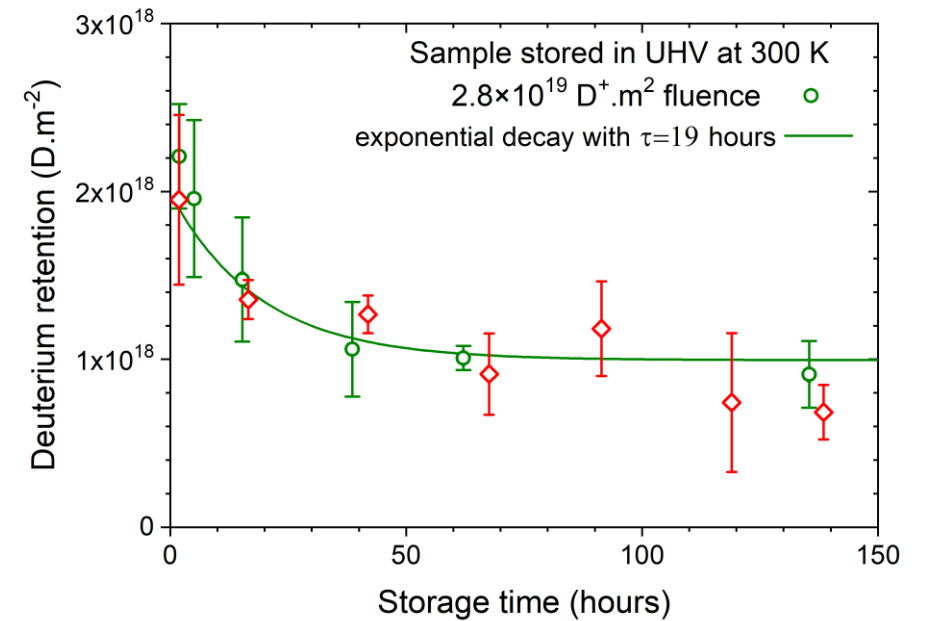
Ialovega *et al.*, in preparation

# Deuterium retention in bulk polycrystalline tungsten oxide

$\text{WO}_3$  – is pure bulk oxide also a trapping layer ?



200 nm thick  
 $\text{WO}_3$  stable up  
to  $\sim 800$  K

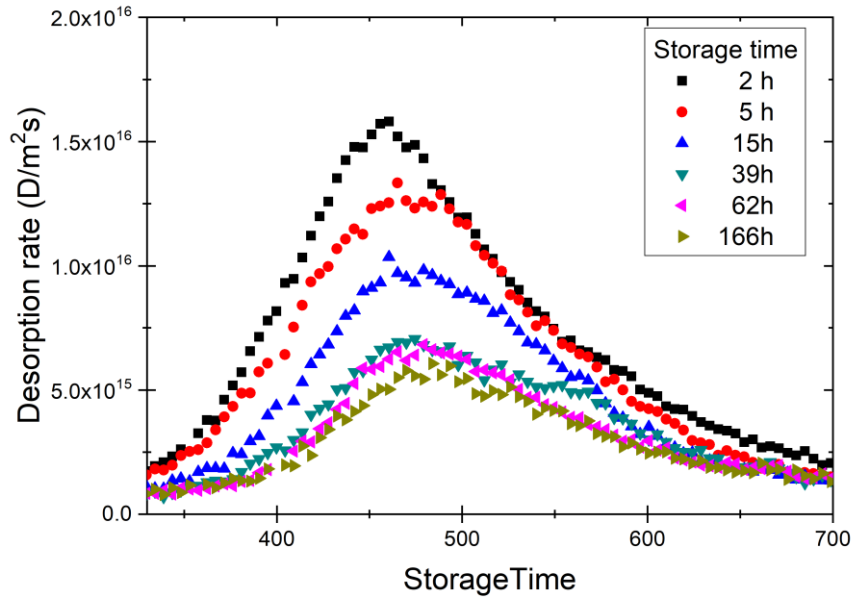


- Low fluence = lower D retention vs poly-W
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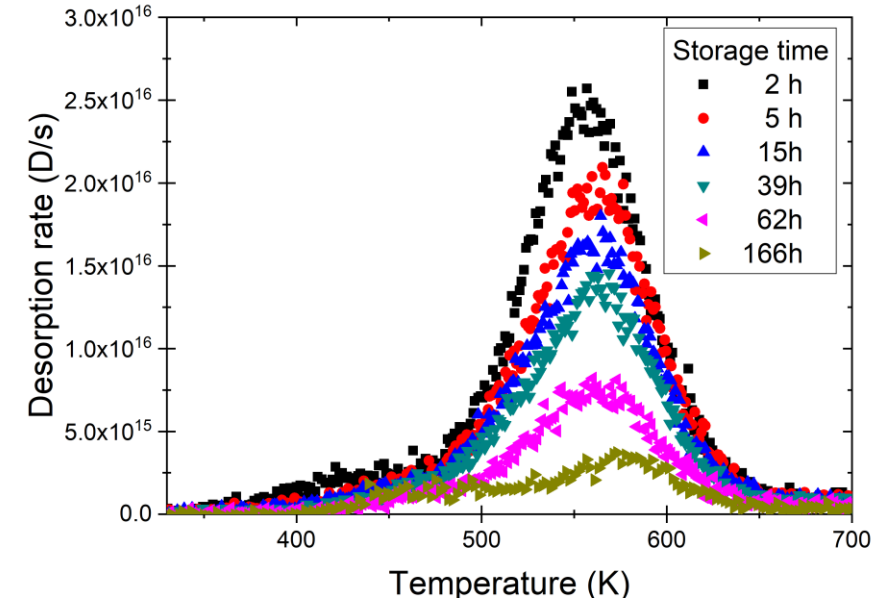
- Isothermal desorption looks similar but...

lalovega *et al.*, in preparation





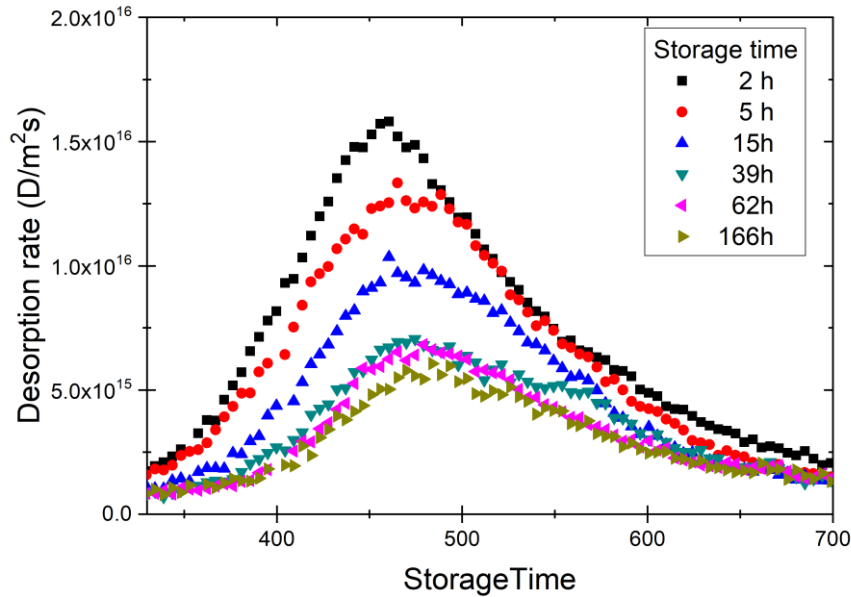
200 nm thick  
WO<sub>3</sub> stable up  
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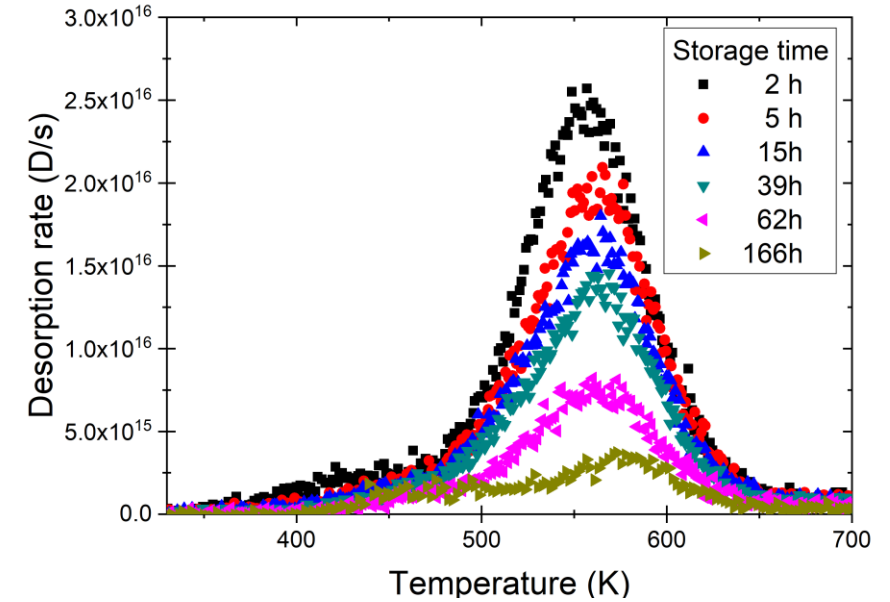
- Low fluence = lower D retention vs poly-W
- High fluence = higher D retention vs poly-W

- Isothermal desorption looks similar but...
- WO<sub>3</sub> decreases down to almost null retention and TPD is really different from poly-W !?

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200 nm thick  
WO<sub>3</sub> stable up  
to ~800 K



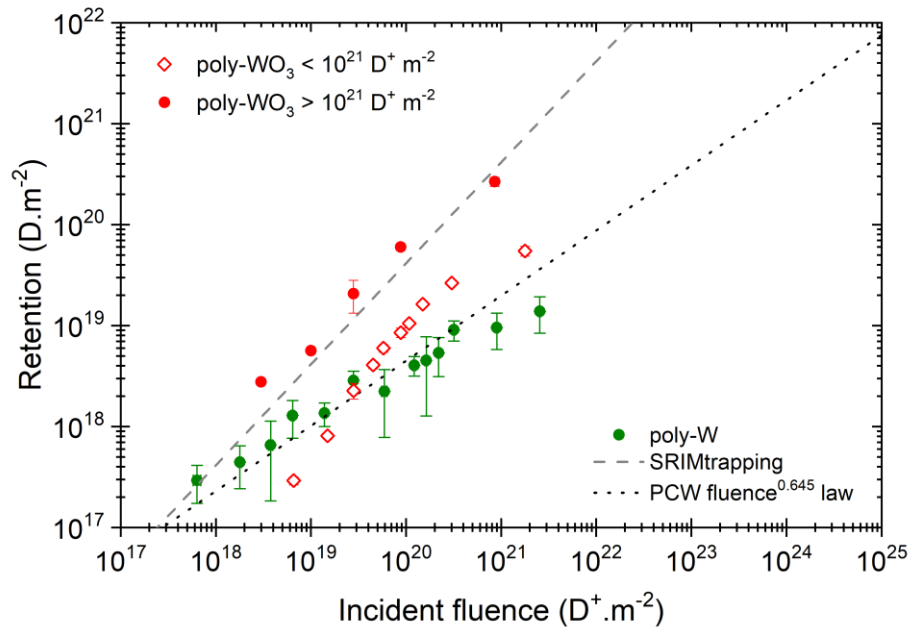
- Low fluence = lower D retention vs poly-W
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Ialovega *et al.*, in preparation

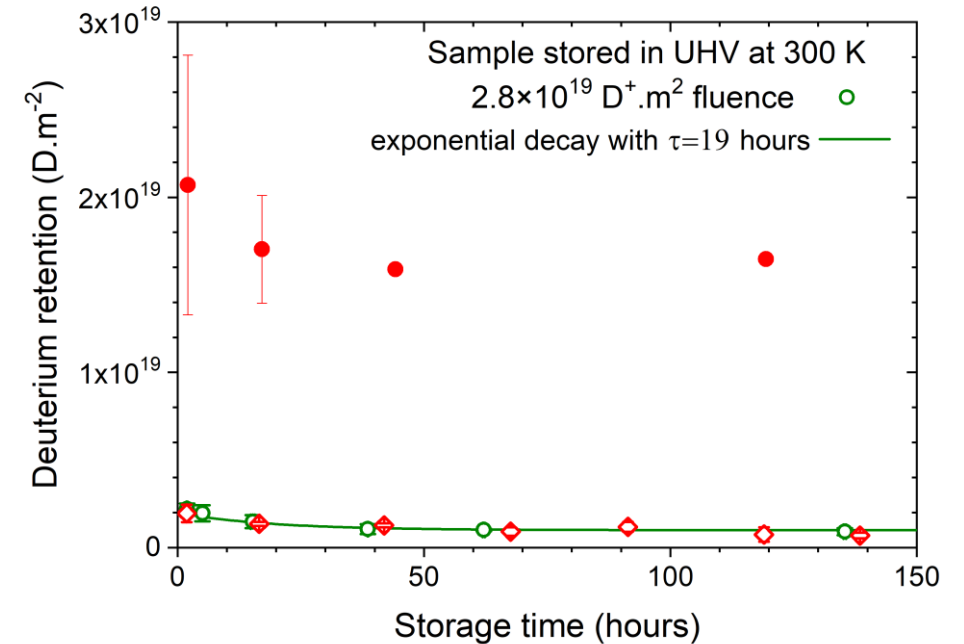
- Isothermal desorption looks similar but...
- WO<sub>3</sub> decreases down to almost null retention and TPD is really different from poly-W !?
- MRE interpretation not straightforward...

# Deuterium retention in bulk polycrystalline tungsten oxide

$WO_3$  – is pure bulk oxide also a trapping layer ?



200 nm thick  
 $WO_3$  stable up  
to  $\sim 800$  K

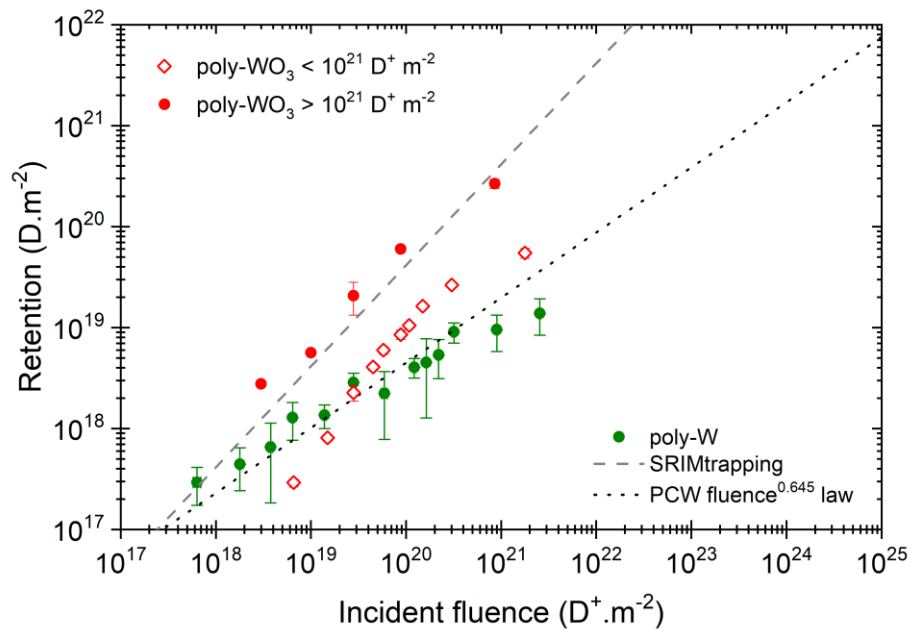


- High cumulated fluence = higher D retention vs poly-W for all implantation fluence

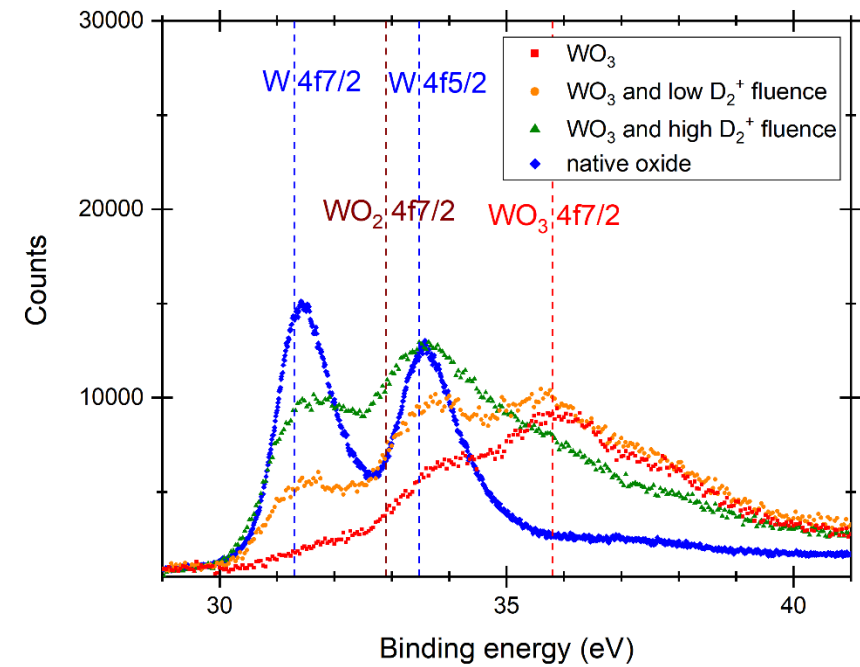
lalovega *et al.*, in preparation

- Isothermal desorption now looks almost negligible...

- MRE interpretation will be complex...
- 3D effects ? columnar structure of  $WO_3$
- Interface effect ? Next slide



200 nm thick  
WO<sub>3</sub> stable up  
to ~800 K



- High cumulated fluence = higher D retention vs poly-W for all implantation fluence

- Increasing cumulated fluence of D<sub>2</sub><sup>+</sup> results in an increasing signature of metallic tungsten at the surface → WO<sub>3-x</sub>

➤ O vacancy in the near surface of WO<sub>3</sub> increases D retention

Ialovega *et al.*, in preparation

- WO<sub>3-x</sub> XPS is still quite different to native oxide's one

# Oxygen atoms on tungsten versus (native) tungsten oxides: effects onto deuterium retention - summary

- W native oxide ( $WO_xC_y$ ) is responsible for some bulk D retention
  - Surface oxygen ( $WO_x$  with  $x < 1$ ) reduces D retention in W (at the surface and in the bulk)
  - Stoichiometric  $WO_3$  reduces D retention at low D ion fluence
  - $WO_{3-x}$  with  $1 < x < 2$  increases drastically D retention at high cumulated fluence
- Isolated O atoms in the bulk of W should explain some of the trapping of hydrogen isotopes in technical tungsten
- Perspective: native oxide = thin  $WO_xC_y$  layer
    - Probing interfaces effect by varying (decreasing)  $WO_3$  layer thickness
    - What about the effect of C on D retention in W ?