

PRD-LMD End-of-year/Kick-off meeting

Droplet ejection from liquid metals

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Sn droplet ejection can lead to unacceptable core radiation

- Droplet emission observed from Sn in several different experiments^{1,2,3}
- Could be strong limit on Sn CPS application due to core contamination
- Not observed for Li samples – why?
- Under what conditions happens, and how to prevent in Sn?

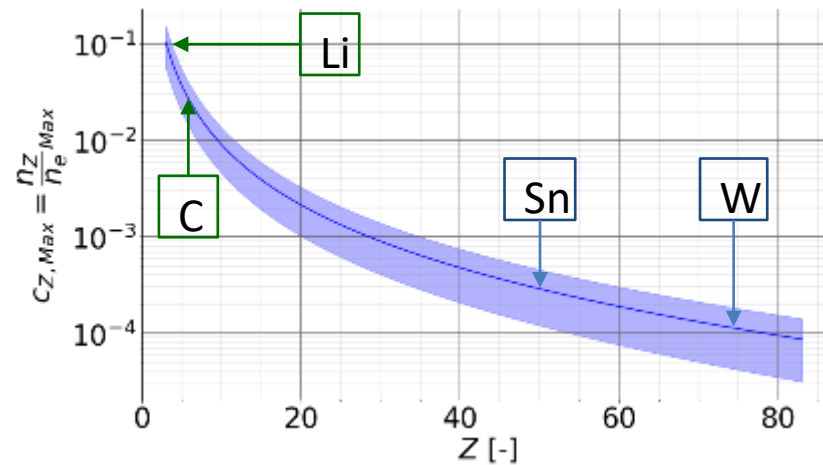
Credits to: Wei Ou



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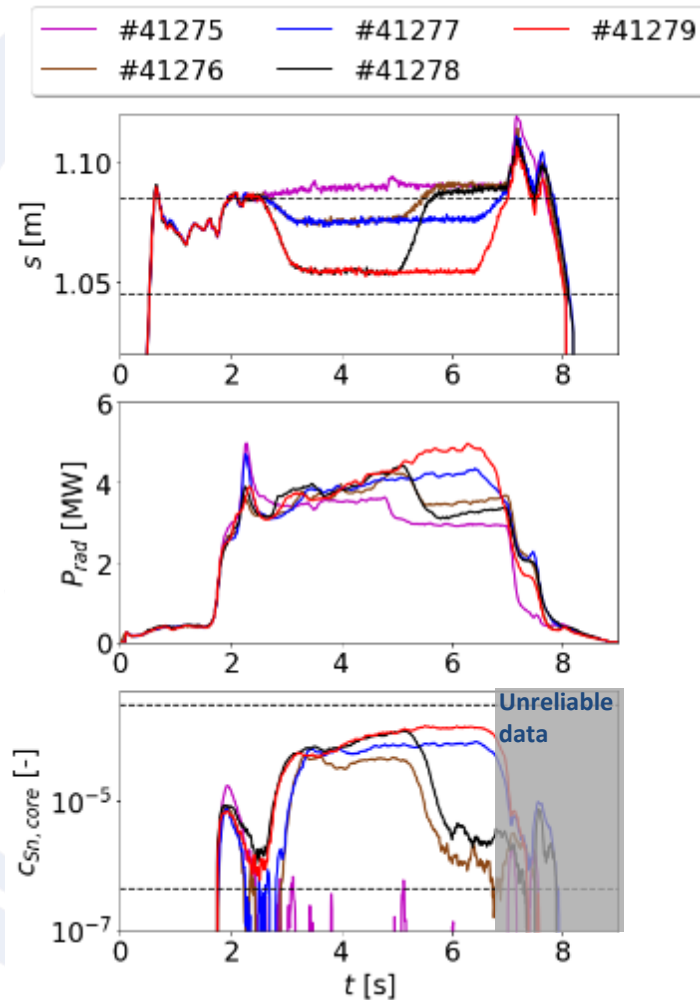
¹A. Manhard *Nucl. Fusion* 60 (2020) 106007
²W. Ou *Nucl. Fusion* 61 (2021) 066030[3]
³J. G. A. Scholte *Nuclear Materials and Energy* 34 (2023) 101315

Maximum impurity concentration
 DEMO for different elements
 Pütterich et al., 2019 NF, 0D model assuming $\tau^*=7,5$



ASDEX Upgrade with an Sn module

1. J. G. A. Scholte et al., *Nucl. Mater. Energy*. **37**, 101522 (2023).





Possible reasons for droplet ejection

1. Localized boiling: Seen in welding (Spatter)

Not possible because:

- Should be more droplet in Li vs Sn
- Droplets are also seen when $T < 500^\circ\text{C}$

2. $\mathbf{J \times B}$ forces:

Not possible because: Nano-PSI (and others) has no B-field

3. Other instabilities:

Unlikely: A CPS should suppress those

3. M. A. Jaworski *et al.*, *Nucl. Fusion*. 53 (2013)

4. T. W. Morgan *et al.*, *Nucl. Mater. Energy*. 12, (2017).

4. Chemical erosion: $\text{Sn} + 4\text{H} \rightarrow \text{SnH}_4 \rightarrow \text{Sn} + 2\text{H}_2$

Unlikely: Surface effect, how does it eject?

5. Hydrogen saturation leading to H_2 bubbles: Champagne

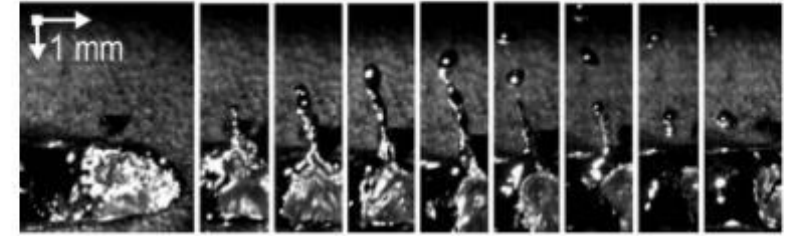


FIG. 2. A high speed imaging sequence ($500 \mu\text{s}$ steps) of droplets being ejected from a laser weld pool. The acceleration of a proportion of the melt in the vertical direction creates a column of melt which later disintegrates into droplets.

2. A. F. H. Kaplan, J. Powell, *J. Laser Appl.* 23 (2011)



Fig. 31 The collapse of hundreds of bubbles at the free surface radiates a cloud of tiny droplets which is characteristic of champagne and sparkling wines and which complements the sensual experience of the taster (© Alain Cornu/Collection CIVC).

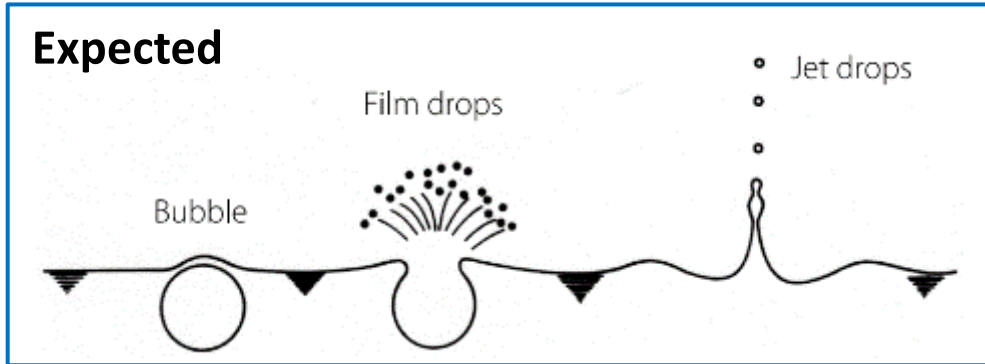
5. G. Liger-Belair, G. Polidori, P. Jeandet, *Chem. Soc. Rev.* 37 (2008).



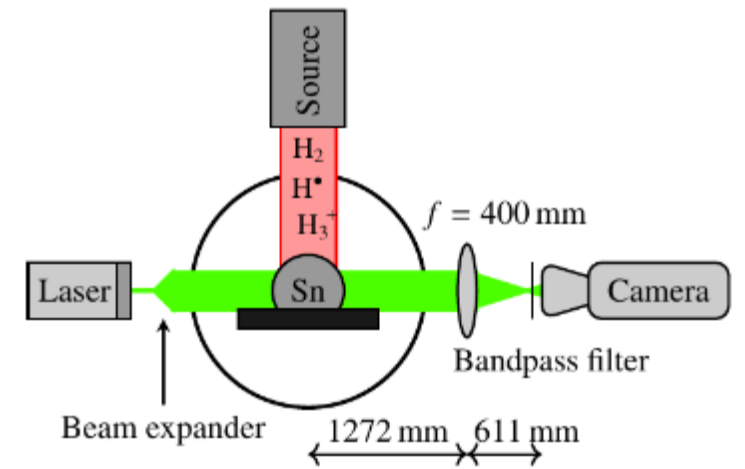
Shadowgraphy

Goal: Test the hypothesis that:

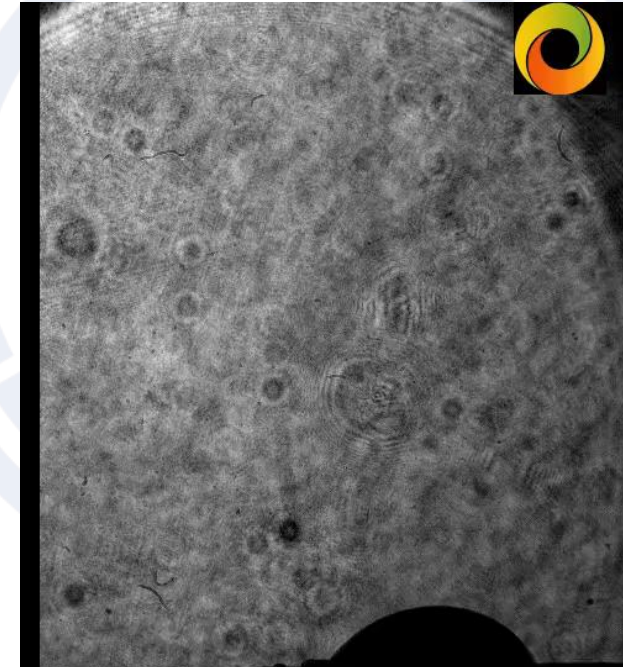
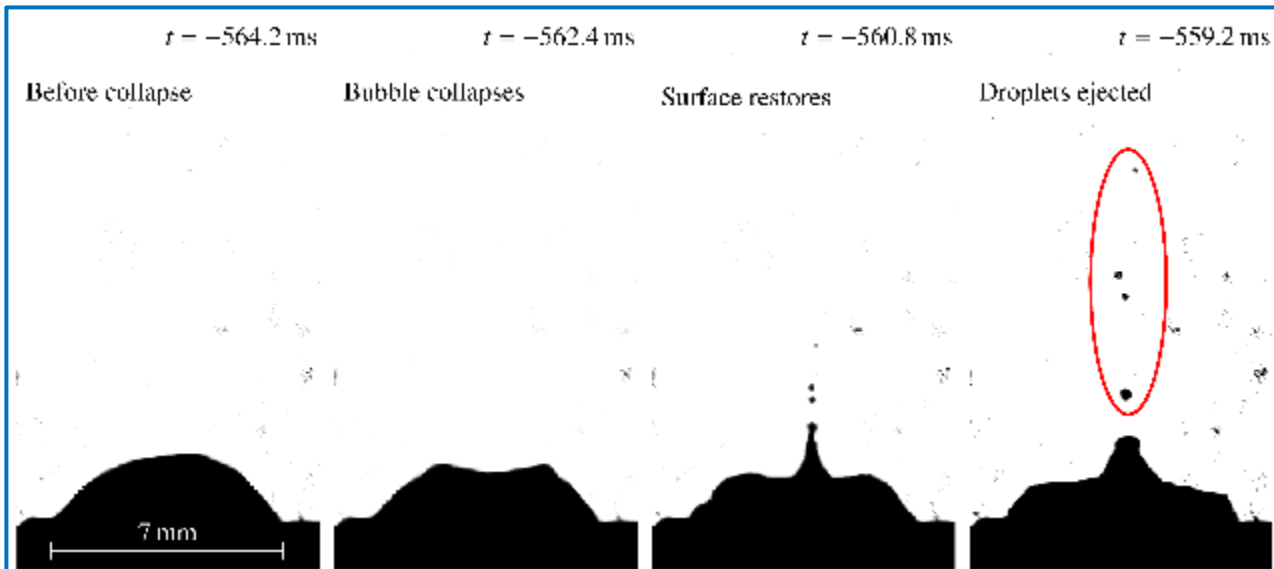
“Sn droplets eject due to a gas bubble collapse”



6. F. J. Resch, J. S. Darrozes, G. M. Afeti, *J. Geophys. Res. Ocean.* **91**, 1019–1029 (1986).



Result



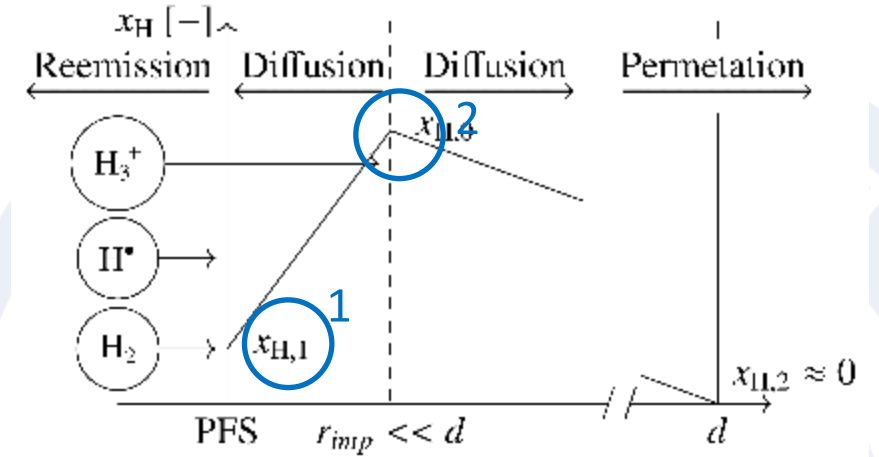
Time: Mon Oct 23 2023 16:26:44.927 884.59, I-22001+:-571.400 ms
Rate: 5000 Exp: 10 μs Durat: 0.115 s



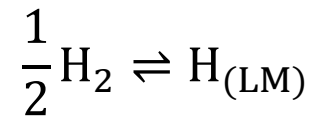
Calculation of hydrogen dissolved in a liquid metal

Calculation plan:

1. Determine x_H on the PFS
2. Determine the increase in x_H in the implantation zone



Reaction equation of hydrogen dissolving in a liquid metal LM

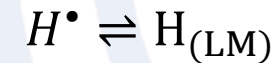


The steady-state mol fraction, using Sievert's law and assuming an ideal gas.

$$x_{H,g} = \frac{\text{Moles of H}}{\text{Total moles}} = K_{H,g} \sqrt{\frac{p_{H_2}}{p^0}}$$

We can find this value in literature

Reaction equation of hydrogen radicals dissolving in a liquid metal LM



$$x_{H,r} = K_{H,r} \frac{p_H}{p^0}$$

$$K_{H,r} = \exp\left(\frac{217,988 - 49,377T}{R_g T}\right) K_{H,g}$$

Independent of the material !!!!!



The influence of radicals and gas

Useful quantities

Saturation ratio: $\alpha = \frac{x_H}{x_{H,g}}$

Supersaturation ratio: $S = \frac{x_H}{x_{H,g}} - 1$

$$S_{H,1} = \exp\left(\frac{217,988 - 49,37T}{R_g T}\right) \frac{p_H}{\sqrt{p_{H_2} p^0}}$$

Laplace pressure bubble: $p = p_0 + \frac{2\gamma}{r}$

Sieverts Law: $\frac{x_{H,crit}}{x_{H,g}} = \sqrt{\frac{p_{crit}}{p_{H_2}}}$

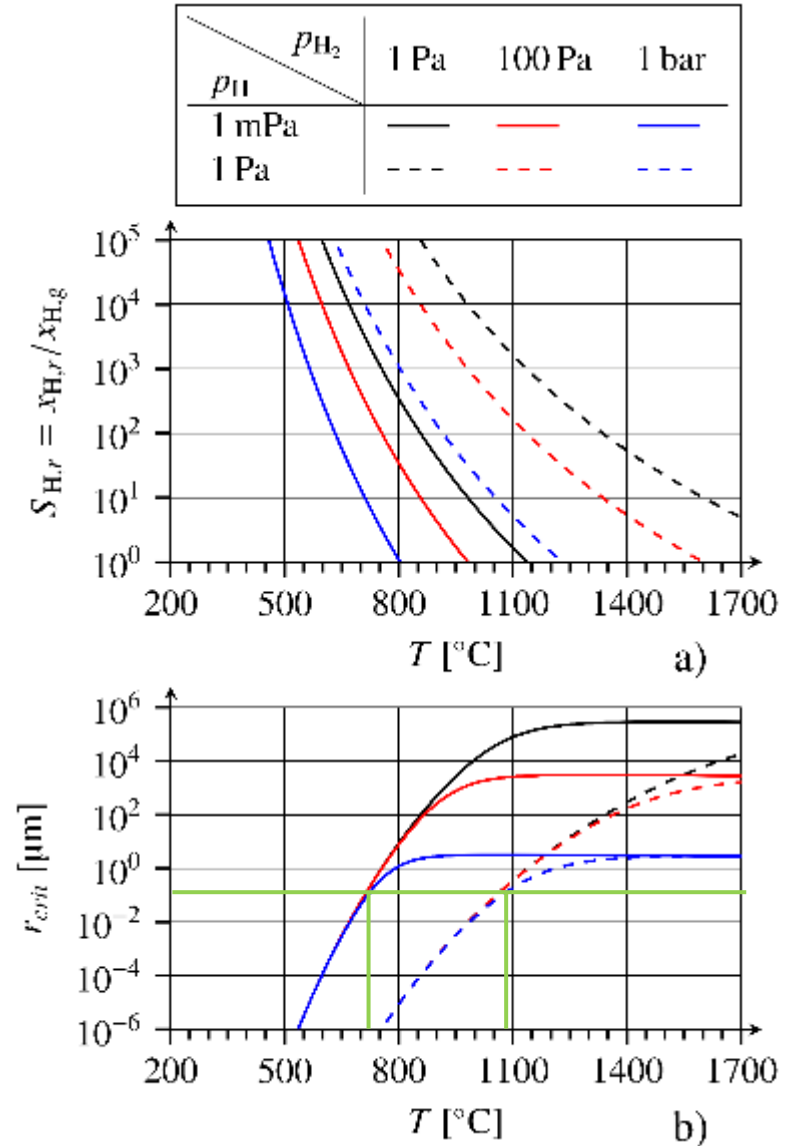
$$r_{crit} = \frac{2\gamma}{p_{H_2}(\alpha^2 - 1)}$$

Conclusion:

- Critical radius depends on the radical partial pressure
- A reasonable CPS will have a pore radius > 0,1µm
 - It could be stable at 750 °C, more realistically 1100°C
- Material independent*

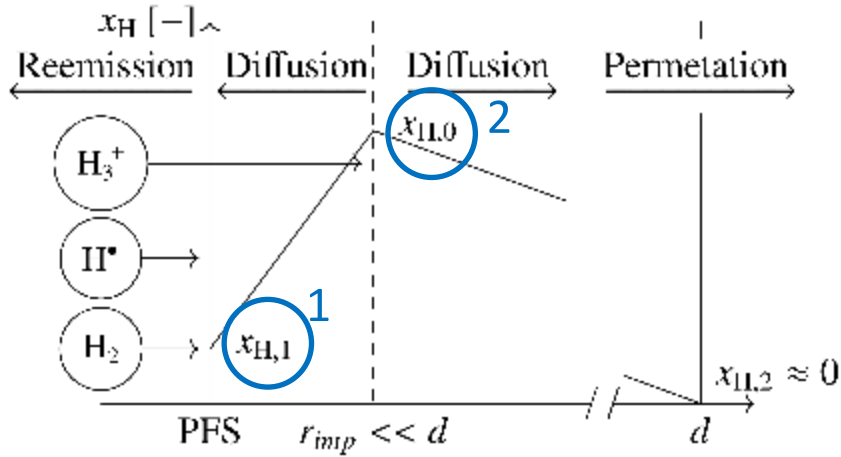
* The critical radius depends on surface tension;
Sn is used in the figure:

	γ [N/m] @750C
Sn	0,51
Ga	0,66
In	0,50
Li	0,25





Step 2 the ions

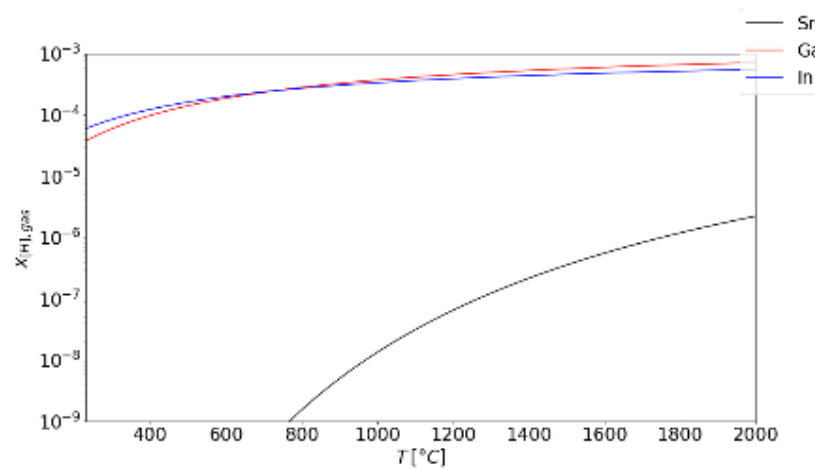


Ficks law of diffusion

$$x_{H,0} < x_{H,1} + \frac{\Gamma_{imp} r_{imp} M_{LM}}{N_A D \rho_{LM}}$$

Ignoring the radicals

$$S_{H,0} = \frac{\Gamma_{imp} r_{imp} M_{LM}}{N_A D \rho_{LM} x_{H,g}}$$

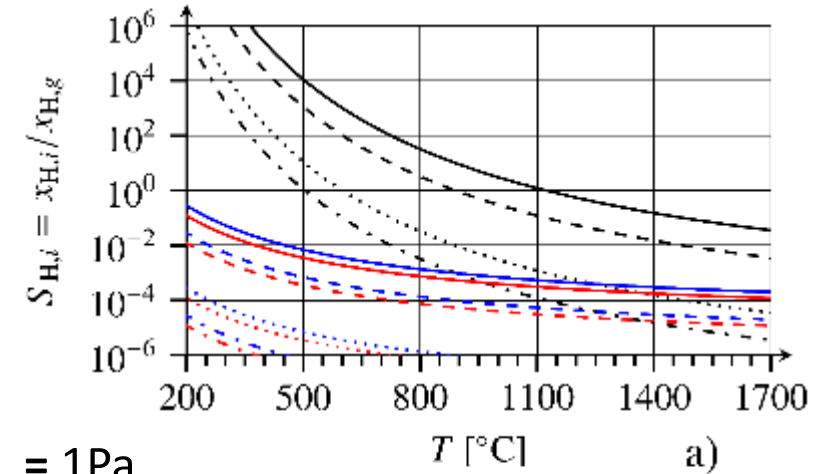


N. D. Deveau, P. S. Yen, R. Datta, *Int. J. Hydrogen Energy*. **43**, 19075–19090 (2018).

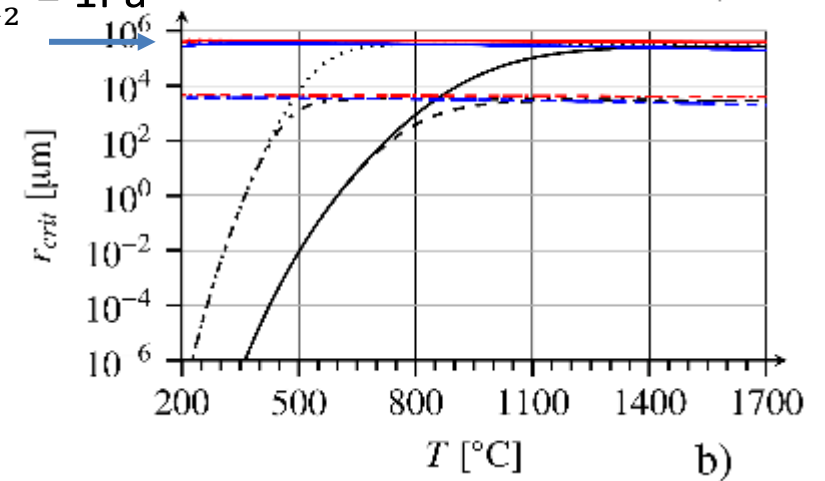
P. W. Humrickhouse, *IEEE Trans. Plasma Sci.* **47**, 3374–3379 (2019).

For Sn, Ga and In

Γ_{imp}	$10^{20} \text{ m}^{-1} \text{ s}^{-2}$	$10^{23} \text{ m}^{-1} \text{ s}^{-2}$
p_{H_2}		
1 Pa	—
100 Pa	-----	----



$p_{H_2} = 1 \text{ Pa}$





Preliminary results in Nano-PSI



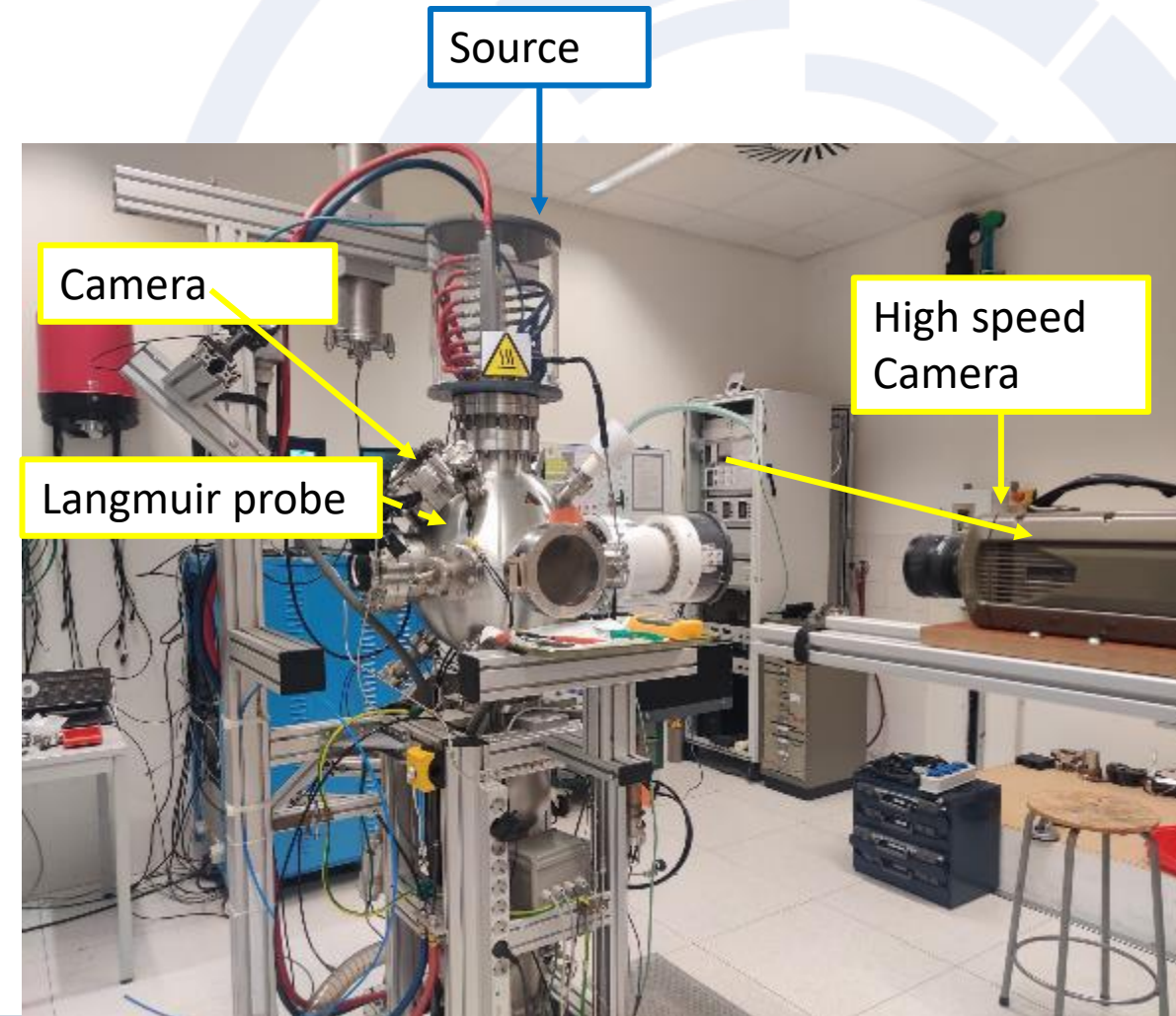


Experimental plan in Nano-PSI

- Expose samples for 30min @ 500°C ±20 °C
- No bias or magnetic field
- Total fluence of $\approx 3,6 \times 10^{23} \text{ m}^{-2}$

Diagnostics:

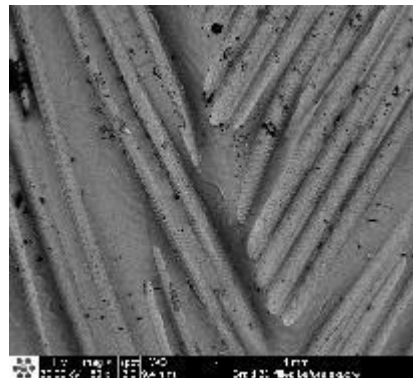
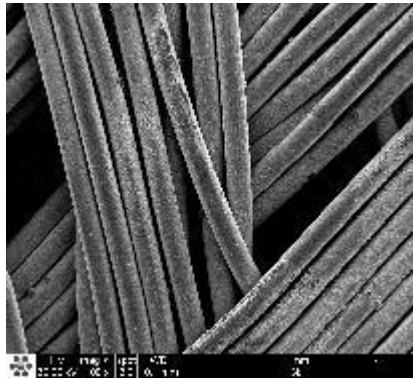
- 4 Stainless steel witness plates
- Thermocouple



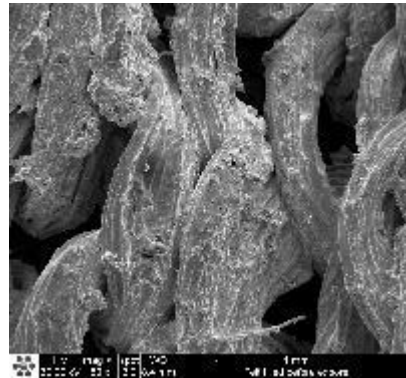


Different CPSs exposed to Nano-PSI

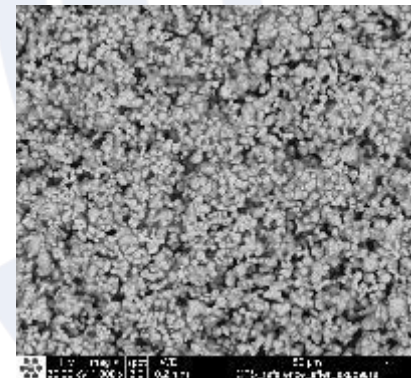
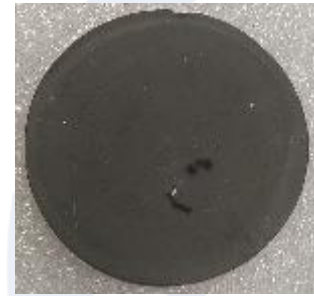
Braids: Hot isolating pressing (UKAEA)



Felts (ENEA)



Sintered disc (Edgetech industries)



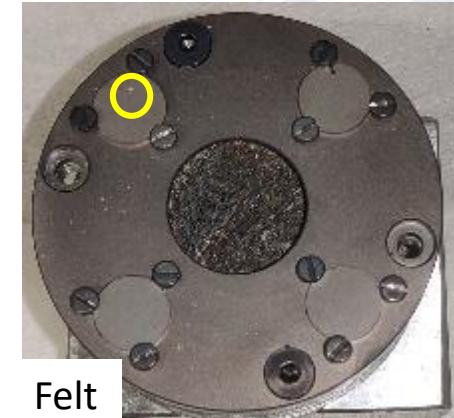
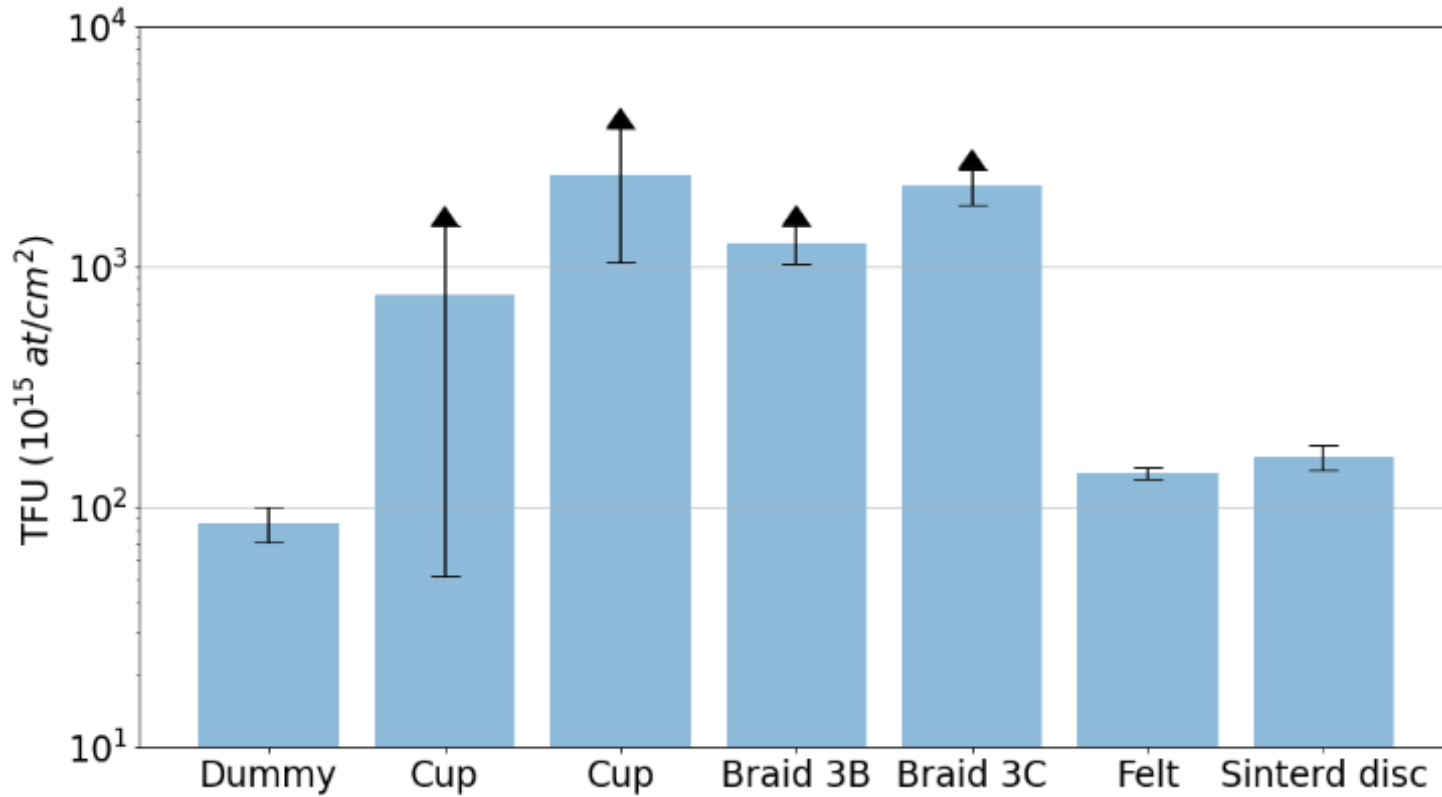


First results on Sn erosion

- Felt and Sintered disc: quite reasonable at droplet suppression

Sample	Mass Loss [mg]
Cup	180,7
Braid 3B	36,3
Braid 3C	108,5
Felt	1
Sintered disc	144,4

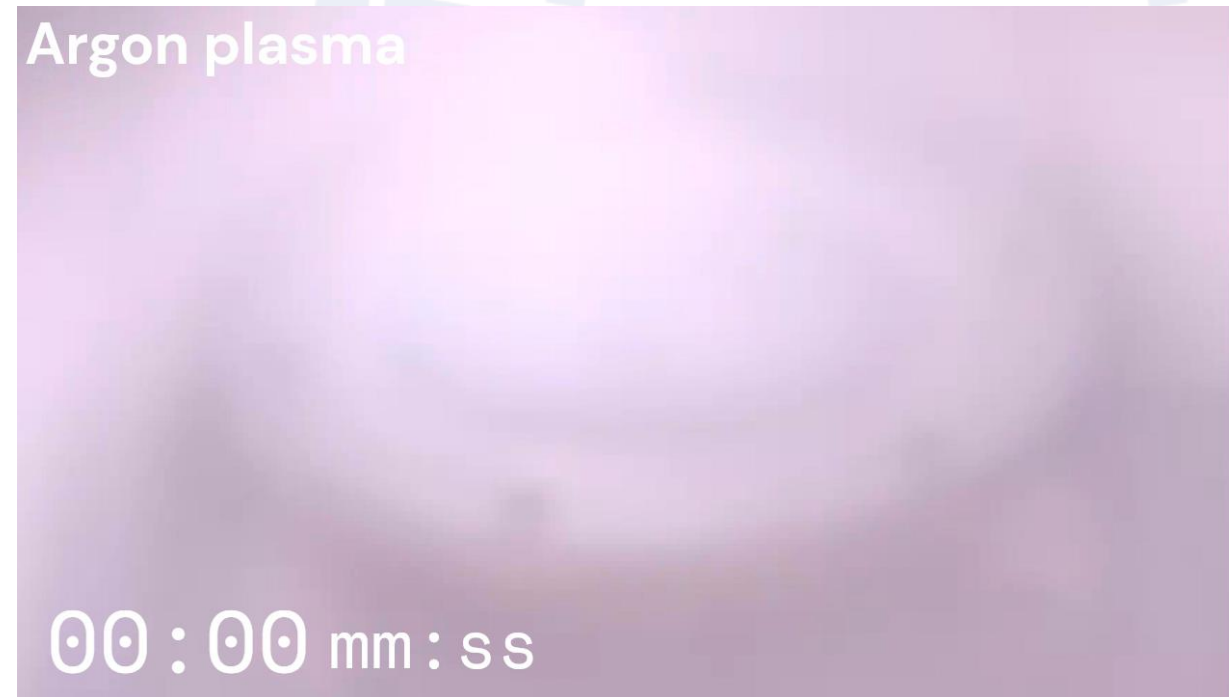
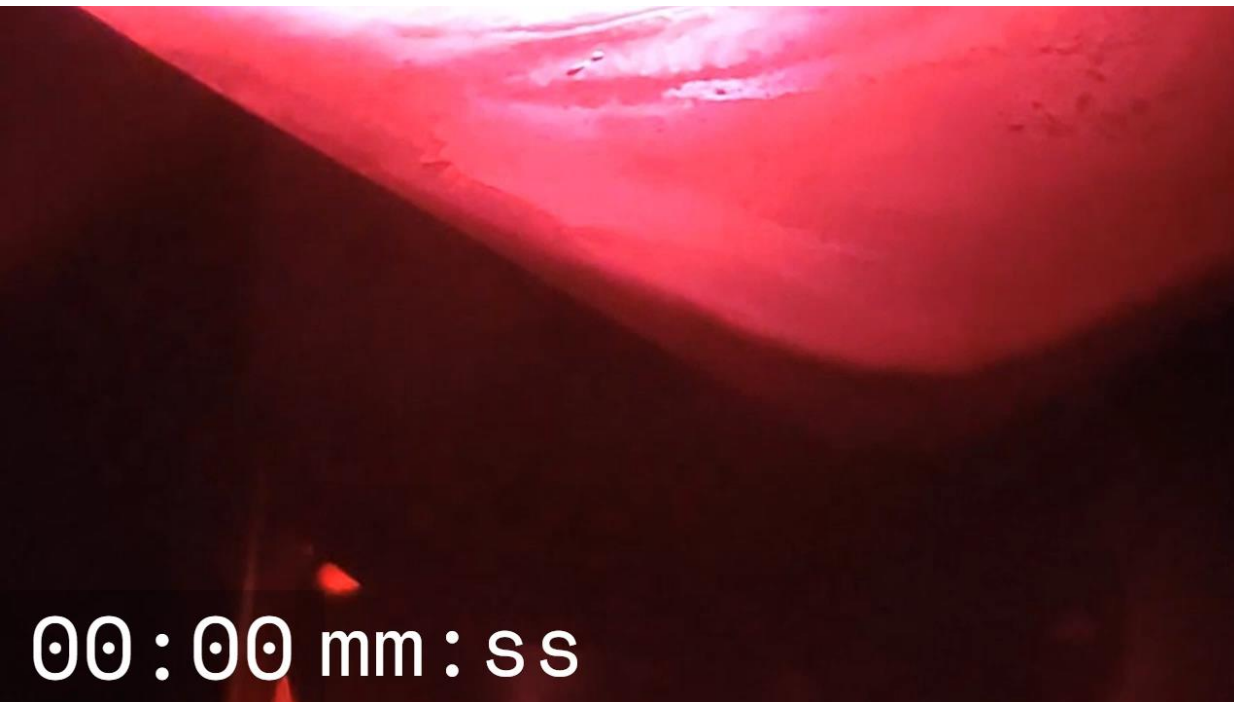
RBS Measurements





Sintered disc “sweats” Sn droplets

- Immediate wettings while switching to hydrogen
- Did not sweat as much with only the heater
- Tin can form a pool on top of the CPS, from which it will spit tin.





Conclusions

- Gas bubble formation is the cause of droplet ejection
- When considering only radicals (and gas)
 - The H solubility in liquid metals are independent of the LM
 - The critical radius is restricted by the radical partial pressure and not the partial gas pressure
- When considering only ions
 - Strong material dependence
 - Sn has an extremely low hydrogen solubility
- With all CPSs Sn droplet ejection, Felt and Sintered disc most promising
- Improvement suggestions various CPS manufacturing techniques
 - Felts/Braids: Thermal connection
 - Sintered: Brittleness, leakage/pooling

To be done in 2024

- What is the influence of ions vs free radicals w.r.t. droplet ejection?
 - Use a radical source with the same flux as in the Nano-PSI.
- Do droplets only eject when using Sn or are other metals also effected
 - Ga and In have a higher solubility → less droplet ejection?
 - Ions vs radial
 - Different pore



Back-up slides



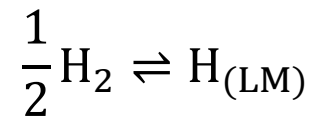


Calculation of hydrogen dissolved in a liquid metal

Calculation plan:

1. Determine x_H on the PFS
2. Determine the increase in x_H in the implantation zone

Reaction equation of hydrogen dissolving in a liquid metal LM



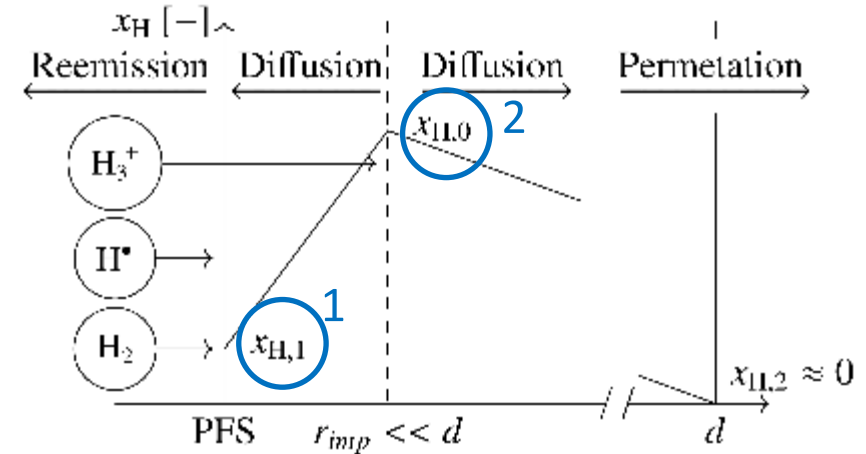
The steady-state mol fraction, using Sievert's law and assuming an ideal gas.

$$x_{H,g} = \frac{\text{Moles of H}}{\text{Total moles}} = K_{H,g} \sqrt{\frac{p_{H_2}}{p^0}}$$

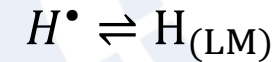
We can write this as:

$$K_{H,g} = \exp\left(-\frac{\Delta G_{H_2}^0}{R_g T}\right)$$

We can find this value in literature



Reaction equation of hydrogen radicals dissolving in a liquid metal LM



$$x_{H,r} = K_{H,r} \frac{p_H}{p^0}$$

$$K_{H,r} = \exp\left(-\frac{\Delta G_H^0}{R_g T}\right)$$

We **cannot** find this value in literature

But we can calculate its value



We can find the reaction constant from first principles

$$\Delta G_{H_2}^O = \Delta G_f^O(H_{(LM)}) - \frac{1}{2} \Delta G_f^O(H_2)$$

$$\Delta G_H^O = \Delta G_f^O(H_{(LM)}) - \Delta G_f^O(H^\bullet)$$

$$\Delta G_H^O = \Delta G_{H_2}^O + \frac{1}{2} \Delta G_f^O(H_2) - \Delta G_f^O(H^\bullet)$$

$$\Delta G_f^O = \Delta H_f^O - T \Delta S_f^O$$

$$\Delta G_H^O = \Delta G_{H_2}^O - 217,988 + 49,377T$$

$$K_{H,r} = \exp\left(-\frac{\Delta G_{H_2}^O - 217,988 + 49,377T}{R_g T}\right)$$

$$K_{H,r} = \exp\left(-\frac{-217,988 + 49,377T}{R_g T}\right) \exp\left(-\frac{\Delta G_{H_2}^O}{R_g T}\right)$$

$$K_{H,r} = \underbrace{\exp\left(\frac{217,988 - 49,377T}{R_g T}\right)}_{K_{H,g}} \rightarrow \text{We can find this value in literature}$$

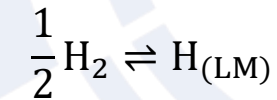
Independent of the material !!!!!

From textbooks one can find that:

$$\Delta H_f^O(H_2) = 0 \text{ Jmol}^{-1}, S_f^O(H_2) = 130,68 \text{ JK}^{-1}\text{mol}^{-1}, \\ \Delta H_f^O(H^\bullet) = 217,998 \text{ kJmol}^{-1}, S_f^O(H^\bullet) = 114,717 \text{ JK}^{-1}\text{mol}^{-1}$$

Reminder

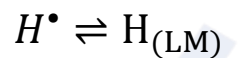
Hydrogen gas



$$x_{H,g} = K_{H,g} \sqrt{\frac{p_{H_2}}{p^O}}$$

$$K_{H,g} = \exp\left(-\frac{\Delta G_{H_2}^O}{R_g T}\right)$$

Hydrogen radicals/atoms



$$x_{H,r} = K_{H,r} \frac{p_H}{p^O}$$

$$K_{H,r} = \exp\left(-\frac{\Delta G_H^O}{R_g T}\right)$$



The supersaturation ratio and critical bubble radius without ions

Useful quantities

$$\text{Saturation ratio: } \alpha = \frac{x_H}{x_{H,g}}$$

$$\text{Supersaturation ratio: } S = \frac{x_H}{x_{H,g}} - 1$$

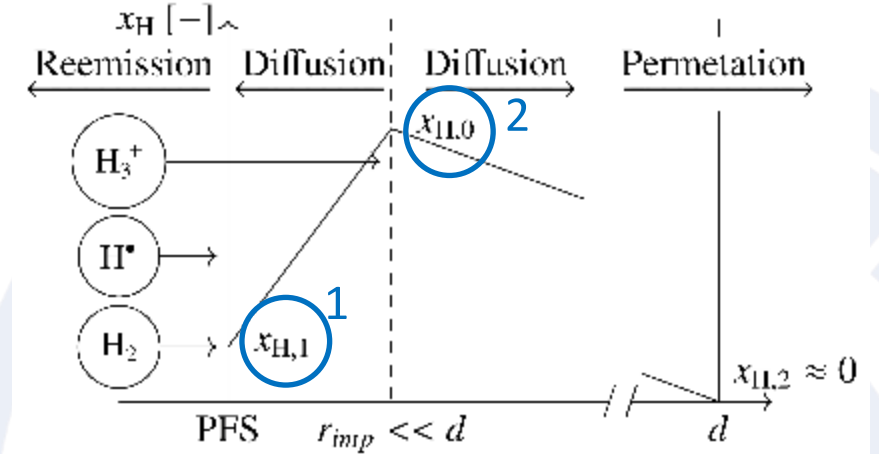
$$\text{Laplace pressure bubble: } p = p_0 + \frac{2\gamma}{r}$$

$$\text{Sieverts Law: } \frac{x_{H,crit}}{x_{H,g}} = \sqrt{\frac{p_{crit}}{p_g}}$$

$$r_{crit} = \frac{2\gamma}{p_g(\alpha^2 - 1)}$$

If a bubble is smaller than r_{crit} the surface tensions will shrink the bubble until it vanishes.

Material independent !!!



Step 1: Determine x_H on the PFS

$$x_{H,1} = x_{H,r} + x_{H,g}$$

$$S_{H,1} = \frac{x_{H,r} + x_{H,g}}{x_{H,g}} - 1 = \frac{x_{H,r}}{x_{H,g}}$$

$$S_{H,1} = \frac{x_{H,r}}{x_{H,g}} = \frac{\exp\left(\frac{217,988 - 49,37T}{R_g T}\right) K_{H,g} \frac{p_H}{p^0}}{K_{H,g} \sqrt{\frac{p_{H_2}}{p^0}}}$$

$$S_{H,1} = \exp\left(\frac{217,988 - 49,37T}{R_g T}\right) \frac{p_H}{\sqrt{p_{H_2} p^0}}$$